Interstitial contribution to self-diffusion in Si: Dependence of the correlation factor on the migration mechanism

related publication:

M. Posselt, F. Gao, H. Bracht:

Correlation between self-diffusion in Si and the migration mechanisms of vacancies and self-interstitials: An atomistic study Phys. Rev. B 78 (2008) 035208



Motivation

self-diffusion via mobile \vee and I:

$$\mathsf{D}_{\mathsf{sd}} = \mathsf{f}_{\mathsf{V}} \, \mathsf{D}_{\mathsf{V}} \, C_{\mathsf{V}} + \mathsf{f}_{\mathsf{I}} \, \mathsf{D}_{\mathsf{I}} \, C_{\mathsf{I}}$$

correlation factors **f**_V and **f**_I:

relation between the mobility of defects and atoms

Compaan and Haven 1956/1958 (statistical theory of diffusion):

diamond lattice: $f_v = 0.5$, $f_l = 0.73$

used so far in almost all papers on self-diffusion in Si



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assumptions

contribution of mobile V:

vacancy mechanism







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Are the values of Compaan and Haven for f_V and f_I really correct?

V. Voronkov 2005

intrinsic point defects in silicon: a unified view from crystal growth, wafer processing and metal diffusion

H. Bracht 2005

simultaneous modeling of dopant (B, P, As) and self-atom diffusion

best interpretation of recent experimental data

by state-of-the-art modeling by assuming

$$f_V = 0.5$$
 $f_I = 0.5 - 0.6$

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determination of f₁ by MD without any assumptionsthis work:about the migration mechanisms

dependence on interatomic potential



Simulation Method

two interatomic potentials: **Stillinger-Weber and Tersoff 3** dynamic simulation of I migration: **5** – **500 ns** (dependent on T)

diffusivity **D**_I

obtained by monitoring the trajectory of the center-of-mass of the I

trajectory is decomposed into n_s segments, for each segment m the squared displacement sd(m) is calculated

$$sd(m) = (\mathbf{R}(t_m) - \mathbf{R}(t_{m-1}))^2$$
 $D_I = \frac{1}{n_s} \sum_{m=1}^{n_s} \frac{sd(m)}{6\Delta t}$

self-diffusion coefficient D_s

sum of the squared displacements of <u>all</u> atoms vs. time

$$\sum_{i=1}^{N} (\mathbf{r}_i(t) - \mathbf{r}_i(0))^2 = const. + 6D_s t$$

D_s: mobility of the lattice atoms due to the presence of a single I



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$$D_{sd} = \mathbf{f}_{V} D_{V} C_{V} + \mathbf{f}_{I} D_{I} C_{I}$$
$$C_{V} = 0: \quad D_{sd} = \mathbf{f}_{I} D_{I} C_{I}$$





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Results

I migration – SW potential

trajectory (1600 K, 4.4 ns)





extended <110> dumbbell



<110> dumbbell



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migration path I: *motion in a {110} plane – translational jump* <*110>dumbbell - <110>extended dumbbell - <110>dumbbell*



(110)

place the cursor on the figures and double click to start the movie

(-110)

movie 2a

1600 K

movie 2b



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migration path II: change between {110} planes – rotational jump <110>extended dumbbell -<110>dumbbell -<110>extended dumbbell

place the cursor on the figure and double click to start the movie



movie 3

1600 K



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static migration barriers







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dynamic migration barriers and diffusivities

defect mobility higher than atomic mobility

 $E_b = 0.98 \text{ eV}$ $E_b = 0.98 \text{ eV}$

$f_I = 0.54 \ (\neq 0.73)$ dumbbell mechanism

(cf. Voronkov, Bracht 2005)





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I migration – T3 potential





<100> dumbbell



<110> dumbbell





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migration path I: *motion in a <100> direction* T - <100>dumbbell - T



migration path II: *motion in a {100} plane T* - <110>*dumbbell* - *T*



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dynamic migration barriers and diffusivities

defect mobility higher than atomic mobility

 $E_b = 0.75 \text{ eV}$ $E_b = 0.75 \text{ eV}$

 $f_I = 0.71 ~(\sim 0.73)$

interstitialcy mechanism

(cf. Compaan and Haven)





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Forschungszentrum Dresden Rossendorf

MD simulations with the SW potential yield a correlation factor f_1 in agreement with the analysis of recent experimental data on defect, dopant and self-atom diffusion

present results on the migration mechanisms should be independent of the absolute value of the self-interstitial diffusivity determined by the MD simulations

self-interstitial diffusion in Si occurs via the dumbbell mechanism

