

**In situ study of the synthesis of thorite (ThSiO<sub>4</sub>) under environmental representative conditions**

Estevenon, P.; Causse, J.; Szenknect, S.; Welcomme, E.; Mesbah, A.; Moisy, P.;  
Poinssot, C.; Dacheux, N.;

Originally published:

July 2020

**Dalton Transactions 49(2020), 11512-11521**

DOI: <https://doi.org/10.1039/d0dt01790f>

Perma-Link to Publication Repository of HZDR:

<https://www.hzdr.de/publications/Publ-30965>

Release of the secondary publication  
on the basis of the German Copyright Law § 38 Section 4.

## In situ study of the synthesis of thorite (ThSiO<sub>4</sub>) under environmental representative conditions

Paul Estevenon,<sup>a,b,†</sup> Jeremy Causse,<sup>\*a</sup> Stephanie Szenknect,<sup>a</sup> Eleonore Welcomme,<sup>b</sup> Adel Mesbah,<sup>a</sup> Philippe Moisy,<sup>b</sup> Christophe Poinsot<sup>b</sup> and Nicolas Dacheux<sup>a</sup>

Received 00th January 20xx,  
Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

Thorite, ThSiO<sub>4</sub> with Zircon structure type, is one of the most abundant natural source of thorium on earth. While actinides are known to form nanoparticles in silicate medium, no direct link between those colloids and crystalline form of thorite was evidenced until now. Here we show that thorite can be produced in experimental conditions close to environmental pH and temperature. Thanks to in-situ Small and Wide Angle X-rays Scattering (SWAXS) measurements, colloids of a few nanometers are first evidenced for low reaction time. These colloids exhibit elongated shapes and finally tend to aggregate after the size has reached 10 nm. Once aggregated, the system goes through a maturation step finishing with the emergence of nanocrystallites presenting thorite zircon structure. This maturation step is longer when the reaction temperature is decreased highlighting kinetic considerations. The conclusions of this article have potential implications in the paragenesis of Th minerals deposits, but also in the behaviour of Th and, by analogy, tetravalent actinides in the environment. The Th-silicate colloids evidenced in this work have, at low temperature and at near neutral pH, a long-term stability and a morphology in favor of a high mobility in groundwaters. If these species are formed in more diluted media, this could be problematic regarding to the spreading of Th and, by analogy of others tetravalent actinides in the environment.

### 1 Introduction

The thorium chemistry may be considered as indicative of the reactivity of the actinide elements at the oxidation state +IV. Due to the abundance of silicate species in the environment, identifying the actinide-silicate interactions is a key point to evaluate the actinides behaviour in environment. On the one hand, thorium, uranium and neptunium are known to form nanometric particles in silicate reactive media<sup>1–9</sup>. The structure, size and stability of these intrinsic actinide-oxyhydroxy-silicate colloids have been investigated<sup>1,3,5,9</sup>. The prevailing size of the colloids is less than 20 nm, they appear as virtually amorphous or poorly structured and exhibit low isoelectric point (pH<sub>IEP</sub> ≤ 4.6)<sup>3,6,8</sup>, corresponding to a negative surface charge. Thus, An<sup>IV</sup> oxyhydroxy-silicate colloids may be able to stabilize significant amounts of An<sup>IV</sup> in a waterborne form and may enhance their mobility in soils and aquifers. On the other hand, the zircon-type structure minerals thorite (ThSiO<sub>4</sub>) and coffinite (USiO<sub>4</sub>) are among the most abundant natural sources of thorium and uranium in ore deposits. Although they could play an important role in the formation of ore deposits, to the best of our knowledge, no specific link between the colloidal species

and the corresponding crystalline phases (i.e. thorite and coffinite) has been highlighted until now.

The thorite formation has been extensively studied using high temperature dry chemistry<sup>10–15</sup> and hydrothermal synthesis methods<sup>13–31</sup>. However, even though thermodynamic measurements established that ThSiO<sub>4</sub> was more stable than a ThO<sub>2</sub> and SiO<sub>2</sub> mixture at 298 K<sup>23,27,32</sup>, hydrothermal conditions or elevated temperatures have always been used to synthesize thorite. To the best of our knowledge, the lowest temperature reported is 150°C in ligand rich reactive media (fluoride or carbonate ions)<sup>16,30</sup> and 160°C in ligand free reactive media (pH = 1 and C<sub>Th</sub> = 4.2 × 10<sup>-2</sup> mol.L<sup>-1</sup>)<sup>31</sup>. Nevertheless, a recent study has demonstrated that increasing the reactant concentration and modifying the pH of the reactive media may enable the formation of the silicate phase on a wider range of conditions<sup>31</sup>. Small and Wide-Angle X-rays Scattering (SWAXS) allowed us to follow the evolution of the size and shape of colloids, to assess the nature of their interactions (electrostatic repulsion, aggregation...) and to identify the nature of crystalline phases formed in solution. This technique was already used to investigate the first stages of formation of various crystalline phases such as zeolite, titanium dioxide, microporous aluminophosphonate, molybdate catalysts, tungstite, zirconia or MOFs. For example, Panzarella et al. showed that microwave synthesis of NaY zeolite started first with the formation of amorphous nanoparticles<sup>33</sup>. Then, these colloids tended to aggregate throughout the synthesis, to finally produce crystalline NaY zeolite. Fan et al. studied LTA zeolite formation and concluded that homogeneous nucleation took place in a first step to produce amorphous nanoparticles of about 5-

<sup>a</sup> ICSM, Univ Montpellier, CEA, CNRS, ENSCM, Bagnols-sur-Cèze, France.

<sup>b</sup> CEA, DES, ISEC, DMRC, Univ Montpellier, Marcoule, France.

<sup>†</sup> Present addresses:

The European Synchrotron, CS40220, 38043 Grenoble Cedex 9, France.

Helmholtz Zentrum Dresden-Rossendorf (HZDR), Institute of Resource Ecology, P.O. Box 510119, 01314, Dresden, Germany.

Electronic Supplementary Information (ESI) available: Figure S1 to Figure S6; Table S1 to Table S3. See DOI: 10.1039/x0xx00000x

15 nm, then a growth step occurred, followed by the crystallization of this species<sup>34</sup>. While in-situ SWAXS technique has been mainly used to study inorganic materials, some authors used it to depict the crystallisation process of Metal Organic Frameworks (MOFs). For example, Stavitski et al. have used SWAXS to differentiate the mechanisms of nucleation /crystal growth for two different MOFs (MIL-53 and MIL-101) obtained from the same molecular precursors but in various solvents<sup>35</sup>.

Therefore, in-situ SWAXS is relevant to study the first moments of a crystallisation process. However, this technique has never been used to describe the crystallization of actinide-silicate phases. To the best of our knowledge, only Neill et al. have used SAXS technique in order to determine the size of uranium silicate colloids and to investigate their stability with regard to pH and concentration<sup>1</sup>. They combined SAXS and EXAFS to propose a core-shell model of the colloids with a core made of UO<sub>2</sub> and a shell made of uranium silicate. However, no direct link between the existence of these colloids and the formation of the crystalline phase, coffinite was made.

Thermodynamic data available for thorite indicated that this phase was more stable at 298 K than the mixture of the binary oxides. Thus, hydrothermal conditions were not required to precipitate thorite from a thorium and silicate solution. However, elevated temperature increased the kinetics of crystallization of thorite<sup>30,31</sup>. The present paper aims to evidence the formation of Th-based colloids in silicate solutions at low temperatures and under environmental conditions. The relevant synthesis conditions were identified thanks to the multiparametric study of ThSiO<sub>4</sub> reported in a recent article<sup>31</sup>. The impact of the temperature and of the pH of the synthesis medium on the kinetics of crystallisation of thorite was investigated using in-situ SWAXS technique. This technique, associated to X-rays diffraction, Raman and IR-spectroscopies allowed to establish the link between the presence of thorium silicate colloids in synthesis solutions and the formation of the crystalline thorite.

## 2 Materials and methods

### 2.1 Sample preparation

All of the reagents used for the syntheses were of analytical grade and were supplied by Sigma-Aldrich. Th(NO<sub>3</sub>)<sub>4</sub>·4-5H<sub>2</sub>O is known to be a highly hygroscopic salt. Therefore, in order to accurate control the thorium concentration, a stock solution was prepared by dissolving thorium nitrate salt in 1 mol·L<sup>-1</sup> HNO<sub>3</sub>. The obtained solution was titrated by inductively coupled plasma atomic emission spectroscopy (ICP-AES) and the thorium concentration obtained reached  $C_{Th} = 0.96 \pm 0.01$  mol·L<sup>-1</sup>. In such acidic conditions, thorium was also protected against hydrolysis.

The synthesis of ThSiO<sub>4</sub> was performed using the protocol we recently developed for preparation of the thorium silicate in carbonate free reactive media<sup>31</sup>. An aqueous silicate solution

was prepared by dissolving Na<sub>2</sub>SiO<sub>3</sub> in water and a precise volume of the thorium stock solution was added to this solution to obtain a reacting mixture with  $C_{Si} \approx C_{Th} = 0.21$  mol·L<sup>-1</sup>. In order to obtain thorium oxide-free samples, a silicate excess of 3 mol.% in the reacting mixture was maintained. The pH was then adjusted to the chosen final value with 8 mol·L<sup>-1</sup> NaOH.

In order to perform in-situ SWAXS measurements, a small volume of the solution was taken and placed in glass capillary tubes (1 mm diameter). These capillaries were then sealed thanks to an epoxy resin which was reticulated during 15 minutes under a UV lamp ( $\lambda = 365$  nm). This hermetic sealing was performed to ensure the confinement of the sample and to avoid sample dehydration.

The prepared ex-situ mixtures were introduced in 5 mL Savillex PFA hermetic vials and kept at room temperature or placed in an oven at temperatures equal to 40°C, 60°C or 90°C for a variable period of time (ranged between 50 and 257 days), while SWAXS capillaries were also kept at fixed temperature (40°C or 60°C) thanks to a specific oven. The synthesis conditions of the ThSiO<sub>4</sub> samples are summarized in **Table 1**, in which in-situ experiments refer to SWAXS samples.

At the end of the heat treatment, samples were cooled down to room temperature. Then, the precipitates were separated from the supernatant by centrifugation for 12 min at 14000 rpm, washed twice with deionized water and once with ethanol, and finally dried overnight at 60°C in an oven.

**Table 1.** Synthesis parameters for the ThSiO<sub>4</sub> samples obtained by ex-situ experiments, corresponding time of in-situ SWAXS experiments performed under the same conditions. Labels are then used in Figures 1, S1, S2 and S4 for easier reading.  $C_{Si} = C_{Th} + 3$  mol.%.

Label	pH <sub>ini</sub>	C <sub>Thi</sub> (mol·L <sup>-1</sup> )	Temperature (°C)	Ex-Situ exp. heat treatment (days)	In-Situ exp. heat treatment (days)
(1)			90	50	
(2)			60	51	90
(3)	3	0.21	40	94	90
(4)			RT	257	30
(5)	8		90		
(6)	6	0.21	90	51	
(7)			60	101	90
(8)	6	0.21	40	94	90
(9)			RT	257	30

### 2.2 Methods and characterization

PXRD data were collected on the resulting powders using a Bruker D8 Advance diffractometer equipped with a lynxeye detector and using Cu K $\alpha$  radiation ( $\lambda = 1.54184$  Å) in a reflection geometry (parallel beam). Adapted sample holders were used in order to avoid any potential radioactive contamination. PXRD patterns were recorded between 5 and 100° (2 $\theta$ ) with steps of 0.019° and a total counting time of 2.5 to 3 hours per sample.

Instrumental function was extracted thanks to measurements on pure silicon metal sample. The collected data were refined by the Rietveld method using the Fullprof\_suite package<sup>36</sup>. During the refinements, different profile and structure parameters, such as the zero shift, unit-cell parameters, scale factor, and overall displacement factor were allowed to vary. However, the size of the crystallite has been determined thanks to the Scherrer equation. This method has been preferred to the Rietveld method because Scherrer equation has also been used to evaluate the size of the coherent-domain of ThSiO<sub>4</sub> on the diffraction peaks obtained by SWAXS measurements.

Raman spectra were recorded with a Horiba-Jobin Yvon Aramis spectrometer equipped with an edge filter and a Nd:YAG laser (532 nm) that delivers 60 mW at the sample surface. The laser beam was then focused on a sample using an Olympus BX 41 microscope with a  $\times 50$ LMP objective, resulting in a spot area of  $\sim 1 \mu\text{m}^2$  and a power of 39 mW. Four scans were recorded for each analysed area and a dwell time of 1 to 30 s was used for each spectrum.

FTIR spectra were recorded in the 300 – 4000  $\text{cm}^{-1}$  range using a Perkin-Elmer FTIR Spectrum 100 instrument. Powdered samples were deposited on the surface of an ATR crystal without any prior preparation. The spectra collected under such operating conditions exhibited a resolution of 1  $\text{cm}^{-1}$ . Four scans were performed to average the measurement error.

TEM observations were conducted on small powder samples deposited on a copper grid using a JEOL 1200 EX II electronic microscope, equipped with a QUEMESA (EMSIS) camera. The accelerating voltage of the electron beam was fixed at 120 kV. Thermogravimetric analyses were performed on the ThSiO<sub>4</sub> samples prepared, in order to determine their hydration content, thanks to a SETSYS evolution ATD/ATG analyser. All of these analyses were performed between room temperature and 1000°C under air.

In-situ SWAXS experiments were conducted using a Guinier-Mering setup with a 2D image plate detector. The X-ray source was a molybdenum anode, which delivered a high-energy monochromatic beam ( $\lambda = 0.71 \text{ \AA}$ ,  $E = 17.4 \text{ keV}$ ), providing structural information over scattering vectors  $q$  ranging from 0.25 to 30  $\text{nm}^{-1}$ . Helium flowed between the sample and the image plate to prevent air absorption. A specific oven was used to set temperature at 40 and 60°C. The sample acquisition time was 1800s. Data corrections and radial averaging were performed by standard procedures. The image azimuthal average was determined by FIT2D software from ESRF (France)<sup>37,38</sup>. At the beginning of the experiment, spectra were acquired each 90 minutes, then the waiting time between two measurements was gradually increased. Spectra were recorded throughout the experiment, from the beginning of the heating up to 3 months of heat treatment.

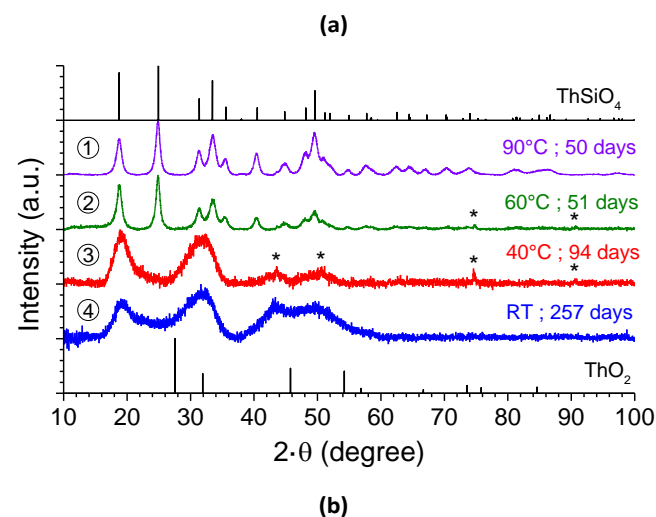
### 3 Results and discussion

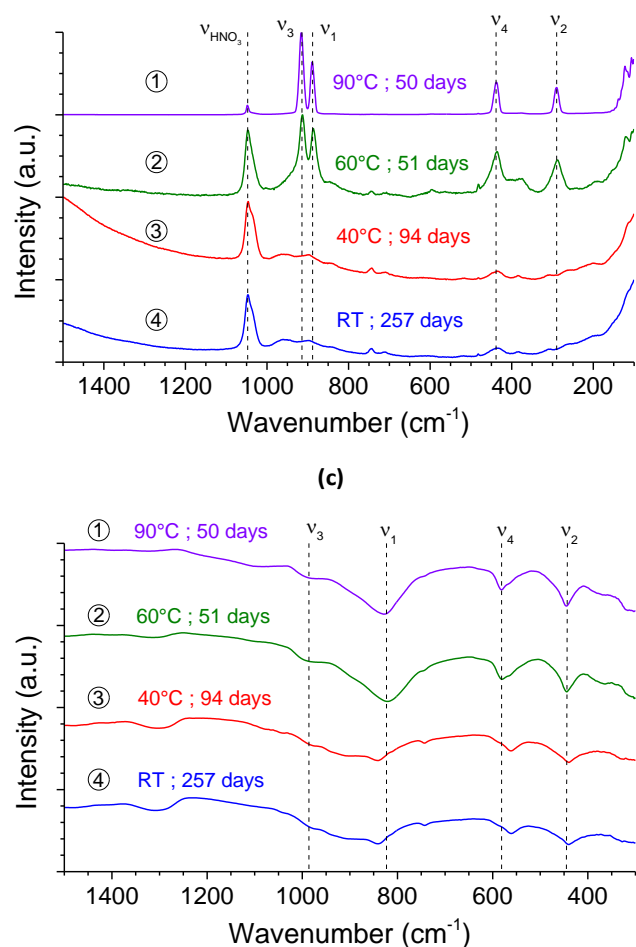
#### 3.1 Characterization of low temperature synthesized ThSiO<sub>4</sub>

Based on the protocol reported for the hydrothermal synthesis of ThSiO<sub>4</sub><sup>31</sup>, several attempts to prepare this phase were conducted at  $C_{\text{Th}} = 0.21 \text{ mol}\cdot\text{L}^{-1}$  and  $C_{\text{Si}} = C_{\text{Th}} + 3 \text{ mol}\%$ , at 90°C for 50 days and pH ranged between 1 and 8. If, the treatment performed at pH = 1 did not lead to the precipitation of any solid phase, the experiment at pH  $\geq 3$  led to the formation of the thorite as single crystallized phase according to the PXRD data (**Figure S1**). The non-precipitation of ThSiO<sub>4</sub> species at pH = 1 is in good agreement with the results we obtained at 120°C and 150°C in our previous study<sup>31</sup>. Moreover, Raman and infrared spectroscopies allowed to confirm the formation of thorite as the final phase obtained without any evidence of the formation of another solid phase (**Figure S1**).

Similar experiments have been performed for pH = 3 and pH = 6 at room temperature, 40°C, 60°C and 90°C for various holding times. Based on the PXRD data, the formation of crystalline thorite has been evidenced at 60°C for pH = 3 (**Figure 1**) and at 40°C at pH = 6 (**Figure S2**). At pH = 3, broad diffraction peaks corresponding to the ThSiO<sub>4</sub> pattern have been observed for the experiments performed at room temperature and 40°C. This behaviour might correspond to the formation of nanocrystalline or poorly crystallized silicate phase. Moreover, the shape of the XRD patterns obtained for the sample synthesized at pH = 6 at room temperature or 40°C suggests also the formation on nanometric or poorly crystallized ThO<sub>2</sub>.

Additionally, the spectroscopic characterization of these samples allowed to confirm the presence of thorite in the final phase (**Figure 1** and **Figure S2**). Indeed, at pH = 3, the formation of thorite has been identified by infrared spectroscopy whatever the temperature considered, especially thanks to the characteristic  $\nu_2$  and  $\nu_4$  bands of the silicate entities.



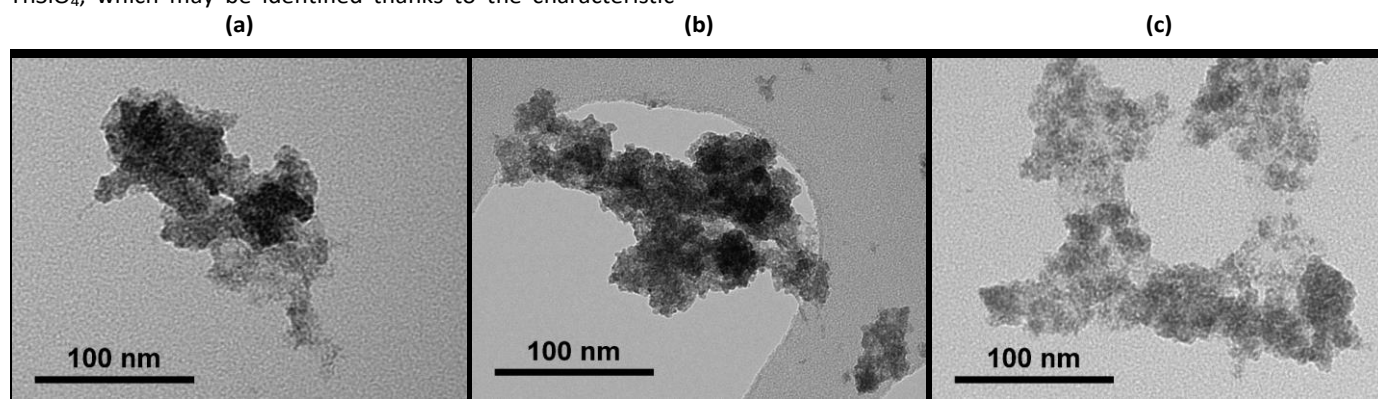


**Figure 1.** PXR patterns (a), Raman (b) and infrared spectra (c) obtained for samples prepared with starting silicate and thorium(IV) concentrations of  $0.21 \text{ mol}\cdot\text{L}^{-1}$  in nitric reactive media at initial pH = 3 at 90°C for 50 days (1), 60°C for 51 days (2), 40°C for 94 days (3) and at room temperature for 257 days (4). Characteristic XRD lines of  $\text{ThO}_2$  and  $\text{ThSiO}_4$  were extracted from references<sup>40</sup> and <sup>3</sup>, respectively.

However, it may be noticed that these bands are shifted for the syntheses performed at low temperature (from  $\nu_2 = 581 \text{ cm}^{-1}$  to  $\nu_2 = 561 \text{ cm}^{-1}$  and from  $\nu_4 = 444 \text{ cm}^{-1}$  to  $\nu_4 = 440 \text{ cm}^{-1}$ ). This shift probably corresponds to a very strong modification of silicate groups geometry at low temperature compared to the ideal  $\text{ThSiO}_4$  structure. Moreover, samples characterization by Raman spectroscopy are also consistent with the formation of  $\text{ThSiO}_4$ , which may be identified thanks to the characteristic

vibration bands associated to the symmetric and antisymmetric stretching modes (at  $892 \text{ cm}^{-1}$  and  $918 \text{ cm}^{-1}$ , respectively), although these bands appeared to be very broadened and attenuated at low temperature. The spectra obtained for the sample synthesized at pH = 6 were consistent with the formation of  $\text{ThSiO}_4$  above 40°C (Figure S2).

Rietveld refinement of the samples synthesized at pH = 3 at 60°C and 90°C allowed to observe a very strong modification of the  $\text{ThSiO}_4$  lattice parameters. Indeed, the  $a$  lattice parameter has been determined to be significantly higher than the reference parameter reported in the literature for thorite single-crystal prepared at 1100°C (at 60°C  $a = 7.1674(18) \text{ \AA}$ ; at 90°C,  $a = 7.1593(9) \text{ \AA}$ ; reference  $a = 7.1328(2) \text{ \AA}$ <sup>39</sup>). Simultaneously, the obtained  $c$  parameter was lower than that reported for  $\text{ThSiO}_4$  single crystal (at 60°C  $c = 6.3081(18) \text{ \AA}$ ; at 90°C,  $c = 6.3099(8) \text{ \AA}$ ; reference  $c = 6.3188(2) \text{ \AA}$ <sup>39</sup>). Consequently, the volume of the lattice was significantly higher for the synthesis performed at low temperature compared to the reference unit cell volume (at 60°C  $V = 324.06(15) \text{ \AA}^3$ ; at 90°C,  $V = 323.42(7) \text{ \AA}^3$ ; reference  $V = 321.48(3) \text{ \AA}^3$ <sup>39</sup>). This behaviour was probably due to the insertion of water or hydroxide groups in the thorite lattice according to the results reported by Frondel et al.<sup>2</sup> and those we obtained recently for  $\text{ThSiO}_4$  synthesis performed under hydrothermal conditions<sup>31</sup>. In order to confirm this hypothesis, thermogravimetric analyses have been conducted up to 1000°C from  $\text{ThSiO}_4$  sample prepared at pH = 3 at 60°C and at 90°C. This analysis allowed to measure a mass loss of 9.85 % and 8.65 % for the samples synthesized at 60°C and 90°C, respectively (Figure S3). These mass losses were principally due to water elimination. It is worth noting that the mass loss observed for the syntheses conducted at 60°C led to the release of 1.97 and 1.71 water molecules per  $\text{ThSiO}_4$  unit for the samples prepared at 60°C and 90°C, respectively. This content matched with the one reported for environmental actinide silicate samples<sup>41–43</sup>. From PXR analysis of the samples heated at 1000°C, the  $\text{ThSiO}_4$  lattice parameters were found to be close ( $a = 7.1337(11) \text{ \AA}$ ,  $c = 6.3259(11) \text{ \AA}$  and  $V = 321.92(9) \text{ \AA}^3$ ) to the reference parameters ( $a = 7.1328(2) \text{ \AA}$ ,  $c = 6.3188(2) \text{ \AA}$  and  $V = 321.48(3) \text{ \AA}^3$ ) (Figure S4).



**Figure 2.** Transmission Electron Microscopy micrographs recorded for the syntheses performed at  $C_{\text{Th}} = 0.21 \text{ mol}\cdot\text{L}^{-1}$ , pH = 3 at 60°C for 51 days (a), at 40°C for 94 days (b) and at room temperature for 257 days (c).

## ARTICLE

Moreover, the formation of a small amount of  $\text{ThO}_2$  (around 10 wt.%) has been observed, probably due to the crystallization of nanometric or very poorly crystallized thorium dioxide or to the degradation of a part of the hydrated thorium silicate when heating.

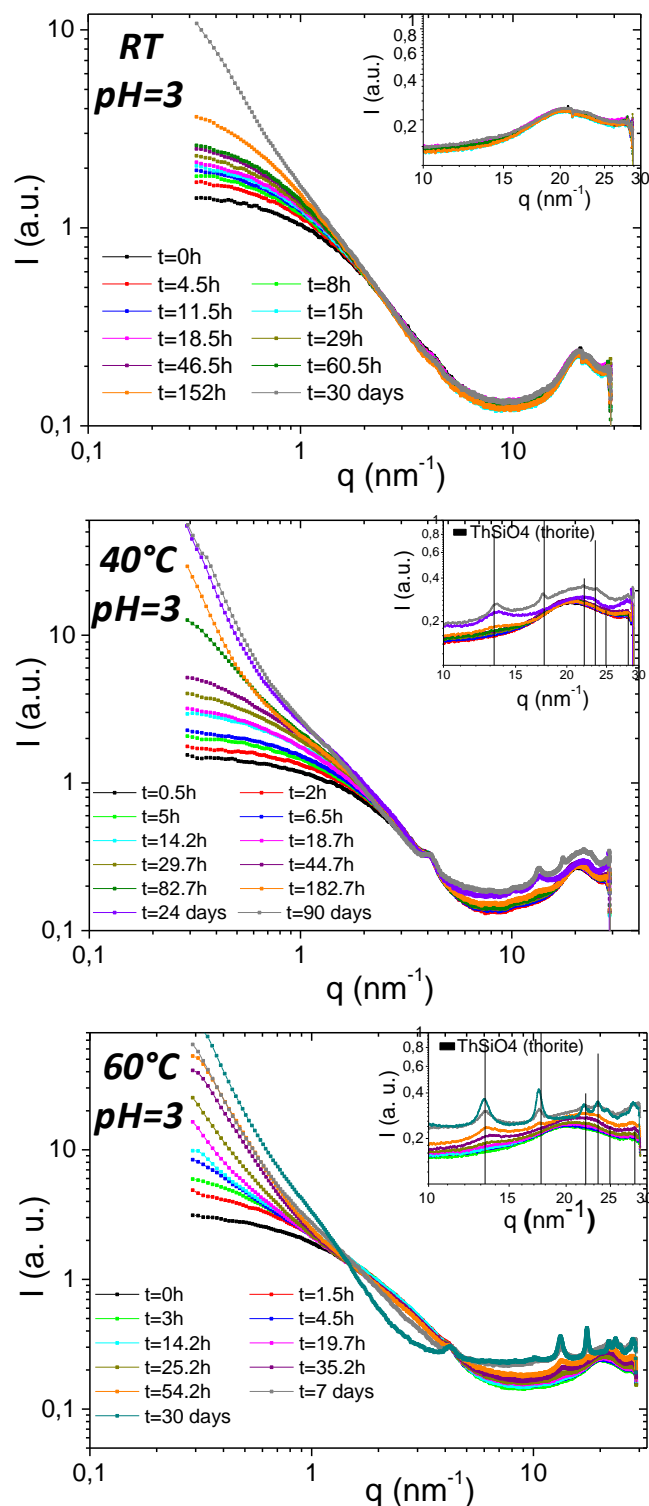
Transmission Electron Microscopy observations have been conducted for various samples (**Figure 2** and **Figure S5**). These experiments did not lead to  $\text{ThSiO}_4$  with a well-defined morphology, but to aggregates composed of small particles of around 10 nm in size. Whatever the temperature of the treatment, difference in the morphology was observed between the samples prepared at pH = 3 and at pH = 6; the first ones being more dispersed than the latest ones.

### 3.2 In-situ study of the $\text{ThSiO}_4$ formation

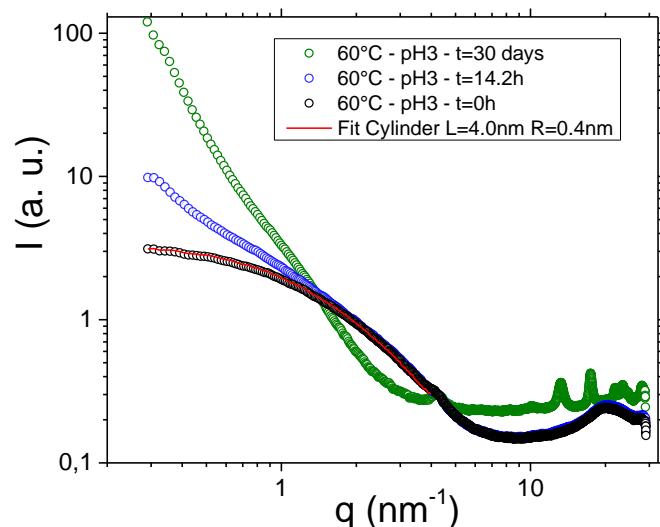
In order to obtain more information on the mechanism of formation of  $\text{ThSiO}_4$ , *in-situ* SWAXS experiments were conducted at pH = 3 and pH = 6 at room temperature, 40°C and 60°C. No experiments were conducted at 90°C due to the brittleness in temperature of the glass capillary tubes used. Complementary experiments were conducted with pH = 1 and pH = 8 at 60°C. However, neither of these experiments actually led to the formation of  $\text{ThSiO}_4$ . Indeed, as previously reported, at pH = 1, no solid phase was found while at pH = 8, the formation of  $\text{ThSiO}_4$  seemed to have been prevented by the precipitation of nanometric or poorly crystallized thorium dioxide particles.

Additionally, it is worth noting that increasing the pH of the reactive media over 3.5 led to the formation of a gel in the reactive media, which conducted to the opacification of the reactive media (from an uncoloured and transparent solution to a white and opaque solution). This precipitation corresponds to the precipitation of particles, which are too big to allow the determination of particle size and shape by SWAXS method. Therefore, the results presented here will mainly correspond to the experiments performed at pH = 3 while the results obtained at pH = 6 are available in supporting information.

SWAXS kinetics curves for three different temperatures at pH = 3 are presented in **Figure 3**. The global examination of these curves indicated first that the systems obtained at pH = 3 exhibited species under the colloidal form for short reaction times. For those samples the SWAXS curves at low  $q$  values showed a quasi-constant variation of the scattered intensity, which characterized the presence of small unaggregated objects. This constant regime at low  $q$  values was no longer observable when the objects started to aggregate. This change in the curve made then possible to set three different steps in the thorite formation.



**Figure 3.** In-situ SWAXS measurements kinetics at room temperature (RT), 40°C and 60°C for pH = 3. The inset graphics refer to a focus in the  $q$  range from  $10\text{nm}^{-1}$  to  $30\text{nm}^{-1}$ .



**Figure 4.** In-situ SWAXS measurements kinetic of thorite formation at pH = 3 at 60°C. Description of STEP I, II and III. At low  $q$  values ( $< 4\text{nm}^{-1}$ ), experiment at  $t_0$  is fitted with a cylinder form factor model,  $P(q)$ .

**Table 2.** Life-times of unaggregated colloids (STEP I); aggregates (STEP II) and for the appearance of  $\text{ThSiO}_4$  crystals (STEP III) for experiments performed at pH = 3, RT, 40°C and 60°C.

	Colloids (STEP I)	Aggregates (STEP II)	$\text{ThSiO}_4$ crystals (STEP III)
pH3 – RT	0 - 152h	152h – 30d	>> 90d
pH3 – 40°C	0 - 44.7h	44.7h – 24d	24d
pH3 – 60°C	0 - 4.5h	4.5h – 54.2h	54.2h

The three steps observed for the system prepared at pH = 3 and 60°C are presented in **Figure 4**. In the first step (STEP I), small unaggregated objects, describable by geometric model such as cylinder, remained in suspension until they started to aggregate (STEP II). As shown on the blue data points series, this aggregation led to the alteration of scattered intensity at low  $q$  values. Then, after a maturation time, crystallites appeared in solution, large enough to give clear diffraction peaks of the thorite crystal (lattice planes (1,0,1) ; (2,0,0) ; (1,1,1) ; (2,1,1) ; (1,1,2) ; (2,2,0)) at large  $q$  values (STEP III). It was therefore possible to assess three different characteristic times for each system (**Table 2** and **Table S1** of supporting information for the samples obtained at pH = 3 and pH = 6, respectively) from the SAXS curves reported in **Figure 3** and **Figure S6**.

In **Table 2**, the time needed to move from unaggregated colloids (STEP I) to aggregated ones (STEP II) decreases when the temperature increases, i.e. from 152 hours at room temperature (20°C) to 44.7 hours at 40°C and to 4.5 hours at 60°C. While constant scattered intensity regime at low  $q$  values for very short lifetime (STEP I) was observed at pH = 3, it was clearly not the case at pH = 6 (**Figure S6**). At pH = 6, the colloids were aggregated from the very beginning of the reaction. Even at room temperature, the scattered intensity at low  $q$  values showed a drastic increase explained by the presence of big objects of several hundreds of nanometers in solution. The

presence of these big scattering objects was attributed to the colloids low surface-charge at pH = 6. Indeed, the isoelectric point of thorium silicate sets at pH = 4.6<sup>3</sup>. In these conditions, electrostatic repulsion do not stabilize colloids and aggregation occurs at the early stage of reaction.

The time needed for the observation of diffraction peaks characteristics of  $\text{ThSiO}_4$  crystals is also presented in **Table 2**. This characteristic time was obtained when crystallites are large enough to give diffraction peaks in the high  $q$  region of SWAXS curves. The thorite peaks appeared for  $q$  values between  $10\text{nm}^{-1}$  and  $30\text{nm}^{-1}$ , in the focus at right bottom of each curve of **Figure 3** (**Figure S6** at pH = 6). As expected, the time needed to detect  $\text{ThSiO}_4$  crystals decreased when temperature increased. At pH = 3 and RT,  $\text{ThSiO}_4$  crystals were not detected, even after one month of reaction whereas  $\text{ThSiO}_4$  diffraction peaks appeared for high  $q$  values after 24 days and 54.2 hours at 40°C and 60°C, respectively. It is worth noting that at room temperature and for pH = 3, crystals of  $\text{ThSiO}_4$  was evidenced by PXRD after 257 days of reaction (**Figure 1a**). At pH = 6 (**Table S1**), only STEP II was observed at RT during 30 days while  $\text{ThSiO}_4$  diffraction peaks appeared for high  $q$  values after 24 days and 35.7 hours at 40°C and 60°C, respectively. Thus, and as expected, at pH = 3 and 6, the kinetics of crystallization of  $\text{ThSiO}_4$  increased with temperature. At geological timescale for example, it is thus reasonable to suggest that those samples should be finally crystallized under the thorite form even at temperature as low as room temperature.

### 3.2.1 From colloidal state to aggregated clusters

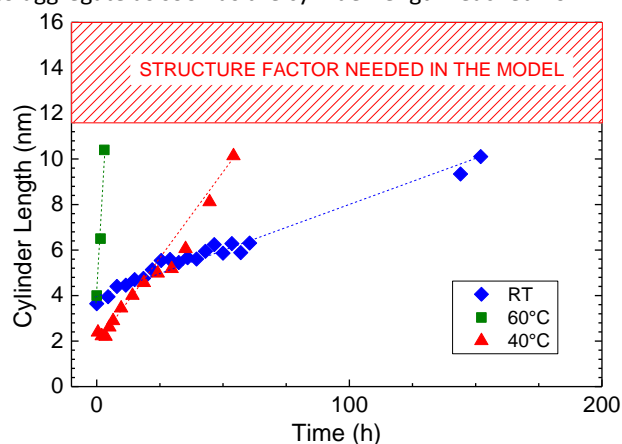
The SWAXS theory claims that the curves relative to scattering objects dispersed in a medium can be described by the following general equation:

$$I(q) = A \cdot P(q) \cdot S(q) \quad (1)$$

Where  $I(q)$  is the scattered intensity;  $A$  is a prefactor depending on scattering objects volume fraction and the molecular volume of objects;  $P(q)$  is the scattering objects form factor; and  $S(q)$  is the structure factor describing interaction between scattering objects.

When the colloids are diluted enough, without any aggregation in solution,  $S(q)$  equals to 1 and the SWAXS curves can be described with only form factor  $P(q)$ . This means that for low reaction times, in the STEP I, the SWAXS curves can be fitted with a simple model not taking into account any structure factor. Therefore, this simplified description was only possible for experiments performed at pH = 3 due to aggregation phenomenon occurring in the gel state at pH = 6. Several form factor models were tested but only cylinder shape exhibited a good correlation with SWAXS curves (see the detailed expression of the form factor model in the supporting information). An example of a fitted curve is presented in **Figure 4** (red curve). Therefore, fitting experimental data points at each reaction time allowed to analyze the evolution of cylinder length  $L$  during STEP I until those cylinders started to aggregate (**Figure 5**). For all the considered temperatures, the cylinder length increased while no noticeable change of the radius was observed (**Table S2**). The growth rate, given by the slope of  $L(t)$ , also increased with the temperature. This result is in agreement

with faster kinetics of reaction at high temperature. It is noticeable that regardless of the temperature, colloids started to aggregate as soon as the cylinder length reached 10 nm.



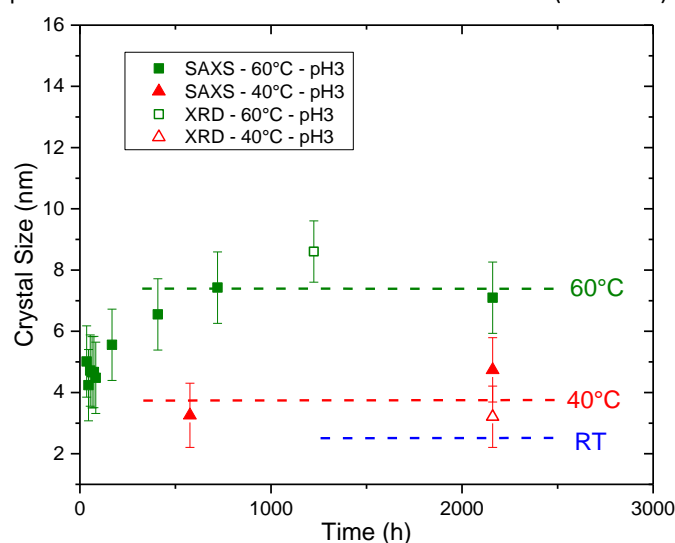
**Figure 5.** Evolution of the cylinders length at low reaction times (STEP I) for SWAXS at pH = 3, at RT, 40°C and 60°C.

It is assumed that at the early stage of the reaction, 1D-growth of the nuclei occurred giving birth to elongated cylinders. Then, at a given time that corresponds to the end of STEP I, these isolated cylinders started to agglomerate leading to larger objects with heterogeneous sizes (STEP II). Another explanation could be that the growth mechanism changed from 1D to 3D. In that case, the extremities of the cylinders would be no more energetically favourable than the walls. This change of reaction site has been already observed for silicates. For example, it is known that silicic acid produced linear structures of silica at low pH values while denser structures were observed for higher pH values due to a change in the localization of condensation reaction in the colloids<sup>44</sup>.

### 3.2.2 Emergence and growth of nanocrystalline domains

If STEP I corresponds to the stage of 1D-growth of colloids and STEP II to an aggregation period, the STEP III refers to the appearance of thorite nanocrystalline domains. The inserts in the curves presented in **Figure 3** show the SWAXS large  $q$  range where thorite diffraction peaks were detected. Thorite crystals were not observed at room temperature for the entire duration of the in-situ SWAXS experiment at pH = 3. However, as already mentioned, PXRD pattern obtained from ex-situ synthesis at RT showed a very broad and diffuse signal after 257 days of reaction, which was the sign of the presence of nanocrystalline thorite. Thanks to ex-situ PXRD and in-situ SWAXS measurements, it was possible to determine the size of coherent domain in thorite crystals using the Debye-Scherrer equation<sup>45</sup>. The variation of the size of the coherent domains in thorite crystals with reaction time is reported in **Figure 6**. Data were treated integrating peak at  $20^\circ$  (PXRD) and  $13 \text{ nm}^{-1}$  (SAXS) corresponding to lattice plane (1,0,1). First of all, the results obtained from SWAXS (full markers) and from PXRD patterns (open markers) are consistent. At RT, diffraction peaks in the large  $q$  range are not observed on SWAXS curves, even after 90 days of experiments. From the results obtained at 60°C, it appeared that the size of  $\text{ThSiO}_4$  crystals increased mainly between 54.2 and 700 hours of reaction, from  $4 \pm 1 \text{ nm}$  to

$7.5 \pm 0.5 \text{ nm}$ . After this first increase, the values reached a plateau until the end of these in-situ measurements (3 months).



**Figure 6.** Evolution of the size of the coherent domains in thorite crystals synthesized at pH = 3 determined from in-situ SAXS and ex-situ PXRD measurements. Dashed lines correspond to crystallite mean size at large lifetime (257 days for RT and pH=3, the data point obtained with PXRD is not represented to ease the reading of the figure).

The dotted blue line in **Figure 6** stands for the size of coherent domains in  $\text{ThSiO}_4$  crystals obtained from PXRD pattern of the synthesis performed at RT after 257 days, i.e. 2.5 nm. After stabilization, the size of the coherent domains in  $\text{ThSiO}_4$  crystals reached  $4 \pm 1 \text{ nm}$  and  $7.5 \pm 0.5 \text{ nm}$  at 40°C and 60°C, respectively. From these results, it appeared that increasing the temperature improved the crystallinity of thorite. It is worth noting that the size of the coherent domains obtained at large reaction time were lower than the maximum length of the cylinders initially formed in the reactive media. This observation indicates that the objects formed at low temperatures (i.e.  $\leq 60^\circ\text{C}$ ) presented a large degree of disorder, which was reduced to some extent by increasing the temperature of the synthesis. Thermogravimetric analyses showed that the amount of water molecules or hydroxide groups in  $\text{ThSiO}_4$  decreased with an increase of the temperature of synthesis. Thus, it could be suspected that a part of the observed disorder in the thorite crystals formed at low temperature (i.e.  $\leq 60^\circ\text{C}$ ) was caused by insertion of water molecules or hydroxide groups in the lattice. The variation of the rate of crystallization with respect to temperature was more difficult to establish due to the lack of values at 40°C and RT.

## 4 Conclusions

For the first time, thorite  $\text{ThSiO}_4$  was obtained from direct precipitation in weakly acid silicate solution at room temperature. This result confirmed the thermodynamic stability at 298 K with respect to the mixture of the binary oxides. In addition, this study allowed following of the early stages of thorite formation. By decreasing the temperature of synthesis in comparison with classical hydrothermal conditions, the kinetic of formation of thorite was considerably reduced. In-situ



SWAXS monitoring of the first steps of crystallization was thus made possible.

This in-situ monitoring allowed characterising of the kinetics of thorite nucleation and growth as a function of the pH value and temperature of the reactive medium. At the early stages of the reaction, SWAXS measurements showed that thorium was maintained in solution in the presence of silicate species under a waterborne form during a long period of time. These Th-silicates colloids adopted a cylinder shape of less than 10 nm length, for a radius around 0.5 nm, before agglomeration or 3D-growth occurred. The duration of this first step increased when decreasing the temperature and pH value. After a given maturation time, during which SAXS signal was no longer describable with simple form factor model due to particles aggregation, the formation of ThSiO<sub>4</sub> crystals was systematically observed by WAXS and PXRD. This result led to the conclusion that colloidal species were either embryo, or intermediate for the crystallisation of thorite. The duration of maturation time was found to increase by decreasing the temperature. The resulting thorite nanocrystals of about 10 nm in diameter were poorly structured. However, the crystallinity of thorite was improved by increasing the temperature. Based on thermogravimetric analyses, it is found that the amount of water molecules or hydroxides groups inserted in the ThSiO<sub>4</sub> crystal structure and responsible for lattice distortion, decreased by increasing the temperature. This behaviour led to less disordered material. Treating these compounds at high temperature (i.e. 1000°C) allowed recovering of the lattice parameters of thorite mineral, while no or very low amount of ThO<sub>2</sub> was observed in the mixtures.

These results have potential implications in the paragenesis of Th minerals deposits, but also in the behaviour of Th and, by analogy, of An<sup>IV</sup> in the environment. Under environmental conditions, concentration of Th based aqueous species is usually several orders of magnitude lower than silicates concentrations due to the very low solubility of Th phases. However, the formation of oxyhydroxide colloids is known to increase the concentration of waterborne Th by several orders of magnitude, even for near neutral pH values. In the presence of a large excess of silicate ions, Th-silicate or Th-hydroxosilicate colloids could exist. The Th-silicate colloids evidenced in this work have, at low temperature and at near neutral pH, a long-term stability and a morphology in favor of a higher mobility in groundwaters. If these species are formed in more diluted media, this could affect the spreading of Th and An<sup>IV</sup> in permeable environmental media, which would allowed colloidal transport.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

The authors would like to acknowledge Bruno Corso (ICSM) for his technical support with PXRD and SWAXS measurements.

## Notes and references

- 1 T. S. Neill, K. Morris, C. I. Pearce, N. K. Sherriff, M. G. Burke, P. A. Chater, A. Janssen, L. S. Natrajan and S. Shaw, *Environmental science & technology*, 2018, **52**, 9118–9127.
- 2 V. Peretroukhine, C. Riglet-Martial, H. Capdevila, V. Calmon, P. Bienvenu and I. Laszak, *Journal of Nuclear Science and Technology*, 2002, **Supplement**, 516–519.
- 3 C. Hennig, S. Weiss, D. Banerjee, E. Brendler, V. Honkimäki, G. Cuello, A. Ikeda-Ohno, A. C. Scheinost and H. Zänker, *Geochimica et Cosmochimica Acta*, 2013, **103**, 197–212.
- 4 C. Riglet-Martial, I. Laszak, L. Bion and P. Vitorge, *Competition carbonate-silicate sur la complexation des actinides en milieu argileux riche en silice*, CEA, 2000.
- 5 H. Zänker, S. Weiss, C. Hennig, V. Brendler and A. Ikeda-Ohno, *Chemistry Open*, 2016, **5**, 174–182.
- 6 I. Dreissig, S. Weiss, C. Hennig, G. Bernhard and H. Zänker, *Geochimica et Cosmochimica Acta*, 2011, **75**, 352–367.
- 7 I. Ulbricht, Ph.D. Thesis. Technischen Universität Dresden, 2010.
- 8 R. Husar, Ph.D. Thesis. Technischen Universität Dresden, 2015.
- 9 R. Husar, S. Weiss, C. Hennig, R. Hübner, A. Ikeda-ohno and H. Zänker, *Environmental science & technology*, 2015, **49**, 665–671.
- 10 L. Troost and L. Ouvrard, *Comptes rendus hebdomadaires des séances de l'Académie des Sciences*, 1887, **105**, 255–258.
- 11 L. H. Fuchs, *The American Mineralogist*, 1958, **43**, 367–368.
- 12 E. R. Vance, *Materials Research Bulletin*, 1986, **21**, 321–329.
- 13 L. H. Fuchs and E. Gebert, *The American Mineralogist*, 1958, **43**, 243–248.
- 14 C. Keller, *Nukleonik*, 1963, **5**, 41–48.
- 15 C. Keller, *Karlsruhe Kernforschungszentrum report*.
- 16 C. Frondel and R. L. Collette, *The American Mineralogist*, 1957, **42**, 759–765.
- 17 D. P. Sinha and R. Prasad, *Journal of Inorganic and Nuclear Chemistry*, 1973, **35**, 2612–2614.
- 18 N. Clavier, S. Szenknect, D. T. Costin, A. Mesbah, C. Poinssot and N. Dacheux, *Spectrochimica Acta Part A: Molecular and Biomolecular Spectroscopy* 118, 2014, **118**, 302–307.
- 19 N. Clavier, S. Szenknect, D. T. Costin, A. Mesbah, J. Ravau, C. Poinssot and N. Dacheux, *Journal of Nuclear Materials*, 2013, **441**, 73–83.
- 20 D. T. Costin, Ph.D. Thesis. Université de Montpellier 2, 2012.
- 21 D. T. Costin, A. Mesbah, N. Clavier, N. Dacheux, C. Poinssot, S. Szenknect and J. Ravau, *Inorganic Chemistry*, 2011, **50**, 11117–11126.
- 22 D. T. Costin, A. Mesbah, N. Clavier, S. Szenknect, N. Dacheux, C. Poinssot, J. Ravau and H. P. Brau, *Progress in Nuclear Energy*, 2012, **57**, 155–160.
- 23 S. Szenknect, D. T. Costin, N. Clavier, A. Mesbah, C. Poinssot, P. Vitorge and N. Dacheux, *Inorganic Chemistry*, 2013, **52**, 6957–6968.
- 24 J. D. Bauer, S. Labs, L. Bayarjargal, W. Morgenroth, S. Weiss, H. Curtius, D. Bosbach and B. Winkler, *DESY annual report*.
- 25 S. Labs, *Schriften des Forschungszentrums Jülich*.
- 26 S. Labs, C. Hennig, S. Weiss, H. Curtius, H. Zänker and D. Bosbach, *Environmental science & technology*, 2014, **48**, 854–860.
- 27 X. Guo, A. Mesbah, N. Clavier, C. Poinssot, D. Wu, H. Xu, N. Dacheux, R. C. Ewing and A. Navrotsky, *Chemistry of Materials*, 2016, **28**, 7117–7124.
- 28 A. Mesbah, N. Clavier, M. J. Lozano-rodriguez, S. Szenknect and N. Dacheux, *Inorganic Chemistry*, 2016, **55**, 11273–11282.
- 29 A. V. Knyazev, M. E. Komshina and I. A. Savushkin, *Radiochemistry*, 2017, **59**, 225–228.

- 30 P. Estevenon, E. Welcomme, S. Szenknect, A. Mesbah, P. Moisy, C. Poinssot and N. Dacheux, *Inorganic Chemistry*, 2018, **57**, 12398–12408.
- 31 P. Estevenon, E. Welcomme, S. Szenknect, A. Mesbah, P. Moisy, C. Poinssot and N. Dacheux, *Inorganic Chemistry*, 2018, **57**, 9393–9402.
- 32 R. D. Schuiling, L. Vergouwen and H. van der Rijst, *American Mineralogist*, 1976, **61**, 166–168.
- 33 B. Panzarella, G. Tompsett, W. C. Conner and K. Jones, *Chemical Physics and Physical Chemistry*, 2007, **8**, 357–369.
- 34 W. Fan, M. Ogura, G. Sankar and T. Okubo, *Chemistry of Materials*, 2007, **19**, 1906–1917.
- 35 E. Stavitski, M. Goesten, J. Juan-Alcaniz, A. Martinez-Joaristi, P. Serra-Crespo, A. V. Petukhov, J. Gascon and F. Kapteijn, *Angewandte Chemie - International Edition*, 2011, **50**, 9624–9628.
- 36 C. Frontera and J. Rodriguez-Carvajal, *Physica B: Condensed Matter*, 2003, **335**, 219–222.
- 37 A. P. Hammersley, *FIT2D: An Introduction and Overview*, ESRF, 1997.
- 38 A. P. Hammersley, *FIT2D V9.129 Reference Manual V3.1*, ESRF, 1998.
- 39 M. Taylor and R. C. Ewing, *Acta Crystallographica Section B-Structural Crystallography and Crystal Chemistry*, 1978, **34**, 1074–1079.
- 40 J. Holzer and G. McCarthy, 1991.
- 41 A. P. Deditius, S. Utsunomiya and R. C. Ewing, *Chemical Geology*, 2008, **251**, 33–49.
- 42 J. Janeczek and R. C. Ewing, *Materials Research Society, Symposium Proceedings*, 1992, **257**, 497–504.
- 43 C. Frondel, *United State Departement of the Interior Geological Survey*.
- 44 R. Corriu and T. A. Nguyễn, *Chimie moléculaire, sol-gel et nanomatériaux*, Ecole polytechnique, Palaiseaux, 2008.
- 45 U. Holzwarth and N. Gibson, *Nature Nanotechnology*, 2011, **6**, 534–534.