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# Formation of nanodiamonds in laser-compressed plastic at planetary interior conditions

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The effects of hydrocarbon dissociation and subsequent diamond precipitation on the internal structure and evolution of icy giant planets like Neptune and Uranus have been discussed for more than three decades<sup>1</sup>. Inside these celestial bodies, gravity compresses mixtures of light elements to densities of several grams per cubic centimeter while the temperature reaches thousands of Kelvins resulting in thermal energies on the order of chemical bonding and above<sup>2</sup>. Under these conditions, simple hydrocarbons like methane, which are highly abundant in the atmospheres of these planets<sup>2</sup>, are believed to undergo structural transitions<sup>3,4</sup> that release molecular hydrogen from deeper layers and may lead to compact stratified cores<sup>5-7</sup>. Indeed, the isentropes of Uranus and Neptune intersect temperature-pressure conditions where first polymerization occurs<sup>8</sup>, and then, in deeper layers, a phase separation into diamond and hydrogen may be possible. Here we show experimental evidence for this phase separation process obtained by *in situ* X-ray diffraction from polystyrene samples dynamically compressed to 150 GPa and 5000 K, which resembles the environment  $\sim 10,000$  km below the surfaces of Neptune and Uranus<sup>9</sup>. Our findings demonstrate the necessity of high pressures for initiating carbon-hydrogen demixing<sup>3</sup> and imply that diamond precipitation may require  $\sim 10$ x higher pressures than previously suggested by experiments investigating non-isolated hydrocarbons<sup>4,8,10</sup>. Besides underlining the general importance of chemical processes inside giant planets, these results will inform evolutionary models of Uranus and Neptune, where carbon-hydrogen demixing can be a significant source for the convection necessary to explain their unusual magnetic fields<sup>11</sup>. Additionally, our experiment demonstrates an alternative path for producing nanodiamonds for scientific and industrial applications<sup>12,13</sup> that may be superior to current methods using oxygen-deficient explosives<sup>14,15</sup>.

Being composed of highly abundant elements, hydrocarbons are one of the most common chemical species throughout the universe. A considerable share exists inside giant planets, especially icy giants like Neptune and Uranus, which are also found in steadily increasing numbers in extrasolar planetary systems. Chemical processes involving hydrocarbons can participate in shaping the interior of these planets and attempts to investigate structural transitions of such systems in the laboratory have been performed by studying methane

inside laser-heated diamond anvil cells and shock experiments with gas guns<sup>4,10,16,17</sup>. In diamond anvil cells, evidence for dissociation and polymerization has been found at pressures of 10-50 GPa and below the carbon melting temperature<sup>4,10</sup>, resulting in a heavy hydrocarbon fluid<sup>8</sup>. Similar structural transitions are expected to happen in various hydrocarbons under shock compression<sup>6,16,17</sup>, which first led to the postulation of possible diamond precipitation inside giant planets<sup>1</sup>. However, due to the lack of *in situ* measurements, details of whether, when and how diamonds can be created are not fully understood.

Some hints of diamond formation have been reported for methane samples in laser-heated diamond anvil cells at pressures of 10-50 GPa and temperatures above 2000 K (Ref. 4) or pressures of 10-80 GPa and temperatures above 3000 K (Ref. 10). Others found the appearance of carbon in form of graphite or diamond at 10-80 GPa and temperatures between 1000 K and 1500 K, but a stable fluid of heavy hydrocarbons at higher temperatures. All this is in contrast to theoretical studies with atomistic simulations, which predict that much higher pressures are required for dissociation and phase separation of hydrocarbons into diamond and hydrogen<sup>3,18,19</sup> ( $\gtrsim 190$  GPa at 2000 K and  $\gtrsim 300$  GPa just above 0 K). Indeed, the diamond anvil cell findings may suffer from an externally forced demixing process since atomic or molecular hydrogen is immediately absorbed by the diamond anvils<sup>20</sup> and the metal gasket surrounding the sample<sup>10</sup>. Moreover, these metal gaskets, as well as pressure standards included in the sample material, are known to significantly lower the temperature threshold for hydrocarbon dissociation<sup>21</sup>. The big differences between the diamond anvil cell results strongly suggest that the observed chemical reactions are mainly influenced by the specific mix of materials in contact with the sample material for each case.

In contrast, dynamic shock experiments on hydrocarbons create isolated systems while still reaching conditions similar to those investigated in diamond anvil cells. In aiming for hydrocarbon dissociation and diamond formation at higher pressures, comparable to the deeper interiors of giant planets, the difficulty is keeping the temperature below the diamond melting line due to the shock-induced entropy increase. Nevertheless, this can be achieved by using a sequence of shocks instead of a single shock, or by applying a ramp compression drive, which can result in a continuous, nearly isentropic compression<sup>22</sup>. Our approach employs a laser-driven two-stage shock compression of polystyrene (CH) up to pressures around 150 GPa, where the temperature is expected to remain below 6000 K. The microscopic structure is then probed *in situ* by ultrafast X-ray diffraction<sup>23</sup>.

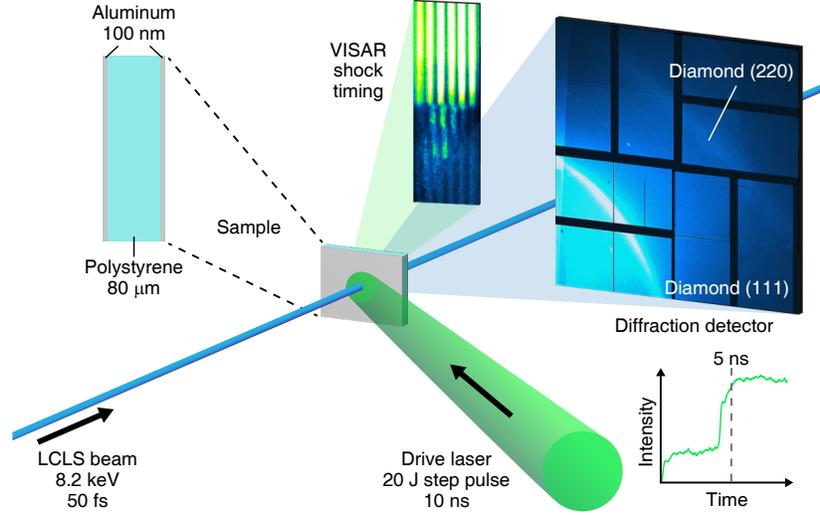


Figure 1. **Schematic of the experimental setup at the Matter at Extreme Conditions endstation of the Linac Coherent Light Source.** Two high-energy laser beams of different intensity and arrival time are overlaid to form a step pulse, which drives two shock compression waves into a polystyrene (CH) film. The shock waves are timed with a VISAR<sup>25</sup> to coalesce at the rear side of the sample. The microscopic state is probed by a single X-ray pulse with 8.2 keV photon energy and 50 fs pulse duration. X-ray diffraction is recorded by a large area X-ray detector, which clearly shows the formation of diamond during the second compression stage.

The experiment reported here was performed at the Matter in Extreme Conditions endstation of the Linac Coherent Light Source of Stanford National Accelerator Laboratory<sup>24</sup>. The available combination of high-energy lasers ( $\gtrsim 10$  J/pulse) with an X-ray free electron laser allows for femtosecond X-ray diffraction in order to directly access the kinetics of shock-induced structural transitions. A schematic of the experiment is shown in Fig. 1. The polystyrene sample is compressed by a step pulse laser profile that had been adjusted for optimal sample conditions using a Velocity Interferometer System for Any Reflector (VISAR)<sup>25</sup> (see Methods section). The initial drive intensity of  $2.7 \times 10^{12}$  W/cm<sup>2</sup> lasts for 4.5 ns before being raised to  $7.1 \times 10^{12}$  W/cm<sup>2</sup>. This results in two stages of shock compression that create conditions similar to planetary interiors at 150 GPa. A single shock compression producing the same final pressure would instead lead to temperatures well above the melting line of diamond and higher than predicted in the interior of the icy giants in our solar system.

One-dimensional hydrodynamic simulations of the two-step shock compression using

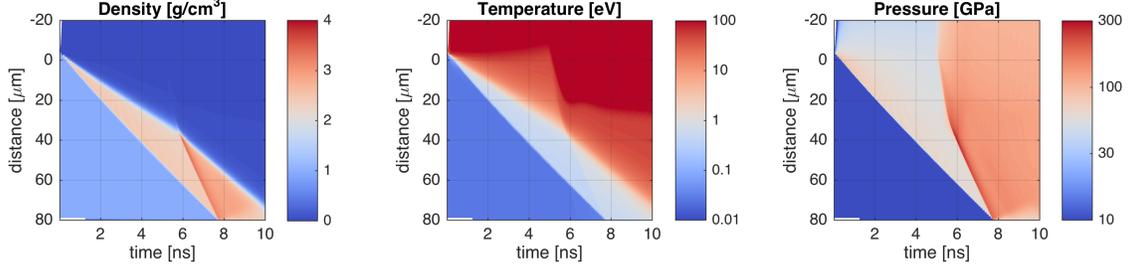


Figure 2. **Hydrodynamic simulations of the two-stage shock compression.** The drive laser hits the sample from the top in all diagrams and the double-stage drive results in two time-delayed shock compression waves. The second shock enters the sample  $\sim 6$  ns after the first compression wave and both shocks coalesce at 7.5 ns when reaching the sample rear side at  $d=80\mu\text{m}$ . At this time, relatively homogeneous conditions of  $\rho=3\text{ g/cm}^3$ ,  $T=5000\text{ K}$  and  $P=150\text{ GPa}$  are predicted in the compressed sample material.

HELIOS<sup>26</sup> are illustrated in Fig. 2. The drive intensity used in the simulations is adjusted to the shock timing measured in the experiment. The first shock has a transit time of 7.5 ns while the second forms when the drive laser reaches maximum intensity and enters the pre-shocked dense material  $\sim 6$  ns after the start of the laser drive. The two shocks then coalesce at the rear side 1.5 ns later. Using the SESAME 7590 equation of state for polystyrene, which gives a good fit to Hugoniot and brightness temperature measurements in this regime<sup>27</sup>, we obtain a density of  $3.0\text{ g/cm}^3$ , a temperature of 5000 K and a pressure of 150 GPa as average conditions in the double-shocked material. This closely resembles the predicted environment at  $\sim 10,000\text{ km}$  into the interiors of Neptune or Uranus<sup>2,9</sup>.

Lineouts of *in situ* X-ray diffraction at various time delays are shown in Fig. 3. At ambient conditions, clear signatures of the amorphous polystyrene are observed together with various weak Bragg reflections from the aluminum coating on both sides. When launching the shock wave into the sample, half of the aluminum coating immediately vanishes as the front layer is vaporized and ablated by the laser drive. At the same time, the ambient amorphous diffraction features start to decrease as the compression waves travel through the sample. In good agreement with the hydrodynamic simulations, which predict that the second shock enters the dense pre-shocked material 6 ns after the start of the laser drive, we find the onset of new diffraction features at 6.2 ns. These peaks are compatible with the (111) and (220) Bragg reflections of compressed diamond at a density of  $4.1\text{ g/cm}^3$  and grow until 7.4 ns,

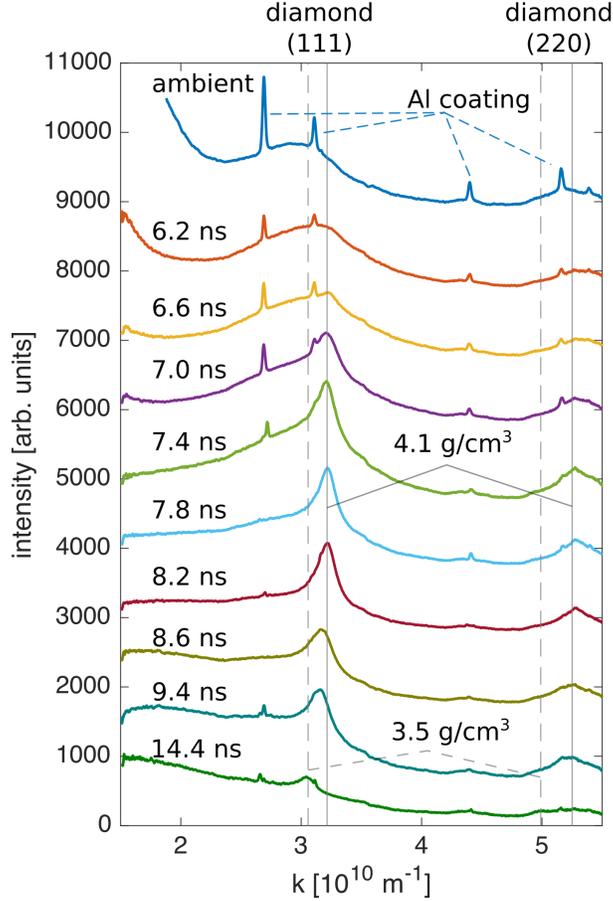


Figure 3. **Diffraction lineouts.** The diffraction signals of cold undriven samples show the typical features of amorphous polystyrene together with Bragg reflections from the aluminum coatings on the front and rear side. After the second shock wave has entered the pre-compressed sample, a strong diffraction signature of compressed diamond starts to appear. This feature remains after the shock release and shows decompression of the nanodiamonds to ambient density.

when the two shock fronts reach the rear surface. The width of the diffraction allows for inferring a lower limit of the nanodiamond particle size via the Scherrer formula<sup>28</sup>. We find a FWHM of  $0.17 \text{ \AA}^{-1}$  (2.9 degrees) for the (111) reflection, which implies nanodiamonds of at least 4 nm in diameter. We do not observe any signature of diamond when just applying a single shock compression drive (shock pressures between 50 GPa and 180 GPa have been applied to the sample). In this case, the compressed polystyrene remains in an amorphous, complex liquid<sup>29</sup> or dense plasma state.

Fig. 4 compares our results to planetary models, atomistic simulations of the carbon-hydrogen phase separation and previous experiments with diamond anvil cells. We do not

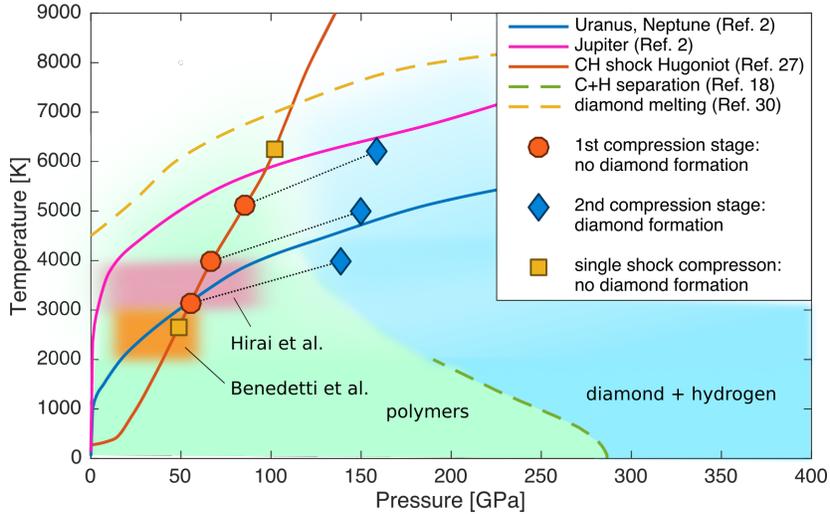


Figure 4. **Summary of the results.** We observe diamond formation around 150 GPa and 5000 K, which overlaps with the predicted isentropes of Uranus and Neptune<sup>2</sup>. At the same time no trace of diamond is found when applying a single shock on polystyrene, which intersects the regime where diamond formation was suggested by experiments in diamond anvil cells (Benedetti et al., Ref. 4 and Hirai et al., Ref. 10). This supports the trend given by atomistic simulations, which predict the phase separation only at higher pressures.

observe any evidence of diamond formation on the probed section of the polystyrene shock Hugoniot curve, which overlaps with the region of the phase diagram covered by experiments with diamond anvil cells. Moreover, we do not find that diamond formation mainly scales with temperature above a pressure threshold of  $\sim 10$  GPa, as reported in Ref. 10. Instead, in our experiment, diamond is only observed after compression by the second shock wave. Diamond is also produced when slightly varying the laser intensity of the two compression stages ( $1.7 \times 10^{12}$  W/cm<sup>2</sup> followed by  $6.4 \times 10^{12}$  W/cm<sup>2</sup> and  $3.2 \times 10^{12}$  W/cm<sup>2</sup> followed by  $8.0 \times 10^{12}$  W/cm<sup>2</sup> were applied). This leads to reduced or increased pressure/temperature conditions: 139 GPa, 4200 K and 159 GPa, 6100 K according to the described hydrodynamic simulations that were benchmarked by the recorded shock timings. In all cases, the second shock raises the pressure by a factor of 2-3, but increases the temperature by only  $\sim 20\%$ . This pressure dependence is more in line with atomistic modeling and therefore suggests that the non-isolated nature of samples tested in diamond anvil cells may obscure conclusions about the carbon-hydrogen phase separation inside planets. In general, our results

support the idea of carbon-hydrogen demixing and subsequent diamond precipitation inside hydrocarbon-bearing giant planets, where the temperature remains below the diamond melting line<sup>30</sup>, but imply that it probably requires higher pressures than suggested by previous experimental work. It needs to be pointed out that the timescales of diamond anvil cell experiments (seconds) are very different to dynamic laser compression experiments (nanoseconds) and both experimental methods are certainly far off quantum molecular dynamics simulations (picoseconds) as well as planetary time scales (millions of years). Therefore, future experiments should aim to obtain precise information on the kinetics of the demixing and phase separation processes in order to achieve extrapolations that can fully constrain models for the internal structure and evolution of giant planets.

In addition to their relevance for planetary modelling, by showing the formation and release of nanodiamonds, our results identify a possible method to produce nanodiamonds from plastics for scientific and industrial applications. The occurrence of so-called detonation nanodiamonds in the soot of oxygen-deficient explosives has been known since the 1960s<sup>14,15</sup> and, since the 1990s, this method has been applied as a commercial source of nanodiamonds, which show a steadily increasing number of scientific and industrial applications<sup>12,13</sup>. Very similar diamond nanoparticles are evidently created in our laser-driven polystyrene samples. In fact, high-repetition rate laser systems (10 Hz or more) with comparable pulse energy as used for our measurements are nowadays readily available and the required double-stage shock compression can simply be realized by using two time-delayed drive lasers (as demonstrated in our experiment). Such laser irradiation of fast moving plastic films above a cooling medium like water could be a cleaner, easier to control and thus eventually cheaper method than the current industrial nanodiamond production using explosives.

## **METHODS**

### **Laser-driven double-stage shock compression**

The samples were laser-cut out of a high-purity polystyrene film ( $\rho_0=1.046\text{ g cm}^{-3}$ ,  $83.4\text{ }\mu\text{m}$  thick) and compressed using two pulsed high-energy lasers (527 nm, 10 ns pulse duration,  $200\text{ }\mu\text{m}$  focal spot diameter, smoothed with random phase plates). The two beams were set to different pulse energies and timing with respect to the X-ray probe in

order to create the desired two-stage shock compression drive. The laser pulse striking the sample first was set to an energy of 8 J, the subsequent pulse reached its maximum intensity 5 ns after the initial pulse and its energy was set to 16 J. The two resulting shock waves were timed with the VISAR system using samples with single-crystal lithium fluoride (LiF) windows at the rear side. With this method, the plastic/LiF interface velocity could be monitored at the moment when a shock front reaches the rear side of the plastic samples. A relatively low interface velocity thus implies that the initial shock reaches the rear side first, whereas a high velocity is clearly connected with the stronger shock overtaking the initial wave before reaching the sample rear side. Adjusting the relative timing and pulse energy of the laser pulses to the values mentioned above ensured shock coalescence very close to the plastic sample rear side. In this way, a reasonably homogeneous sample can be probed by the X-rays at the moment when both shocks reach the rear side. The other described drives were realized with 6 J or 11.5 J in the first pulse and again 16 J in the second pulse. In these cases, the shocks are not optimally timed to coalesce at the rear side, however, clear diamond formation is also recorded starting at the instance when the second shock is expected to enter the dense pre-shocked material. Without the LiF window at the rear side, the VISAR just records a strong drop in reflectivity, but no fringe shift due to rapid evaporation and disassembly of the reflective 100 nm thick aluminum coating. However, during the X-ray shots, the LiF windows were not applied in order to avoid obscuring diffraction signals as well as damage to the X-ray detectors due to strong single crystal reflections. The spatial resolution of the VISAR diagnostic also allows for assessing planarity and the steadiness of the shock waves. We find a planar shock release at the sample rear side in a 50  $\mu\text{m}$  window. Moreover, the polystyrene/LiF interface velocity remains constant within the planar region of the shock release for several ns, which implies reasonably steady shocks. The arrival time of the drive laser was determined by the same streak cameras as used for the VISAR system by scattering light of the drive laser beams from the exact position of sample interaction into the VISAR system.

### **Ultrafast X-ray diffraction**

The polystyrene samples were probed using the LCLS XFEL beam in SASE mode (8.2 keV photon energy, 0.3 % spectral bandwidth, 50 fs pulse duration, 20  $\mu\text{m}$  spot size,  $\sim 3$  mJ per

pulse corresponding to  $2 \times 10^{12}$  photons per pulse). Using imprints of the X-rays in the aluminum coating on the sample rear side that is monitored by the VISAR diagnostic, an accuracy of  $20 \mu\text{m}$  or better could be achieved for the spatial overlap of X-rays and the drive lasers. The timing of the X-ray pulse relative to the drive laser was chosen to probe the sample slightly before or exactly at the moment when the shock waves coalesce at the end of the plastic samples. X-ray diffraction was measured by an  $8 \times 8 \text{ cm}^2$  Cornell-Stanford Pixel Array detector (CSPAD) at  $8.7 \text{ cm}$  distance from the sample covering diffraction angles from  $18$  to  $85$  degrees above the XFEL beam axis. In this way, perturbing effects due to the horizontal polarization of the X-ray beam could be avoided. Angular calibration for the recorded  $2\theta$  range was obtained by diffraction of  $\text{CeO}_2$  and  $\text{LaB}_6$  powder samples.

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## **AUTHOR CONTRIBUTIONS**

D.K., R.W.F., J.V., T.D., A.P., S.H.G., D.O.G., M.R., E.Ga., E.Gr., P.N. and A.J.M. were involved in the project planning. D.K., E.Ga., N.J.H., A.P., T.D., P.N., E.E.M., A.M.S., S.F., L.B.F., P.S., M.J.M., E.J.G., E.Gr., I.N. and T.v.D. carried out the experiment. Experimental data was analyzed and discussed by D.K., J.V., N.J.H., S.H.G., D.O.G., A.P., E.E.M. and T.D. The manuscript was written by D.K., J.V., A.P., N.J.H., M.J.M., D.O.G. and T.D.

## **ADDITIONAL INFORMATION**

The authors declare that there are no competing financial interests. Correspondence and requests for material should be addressed to D.K. (d.kraus@hzdr.de).