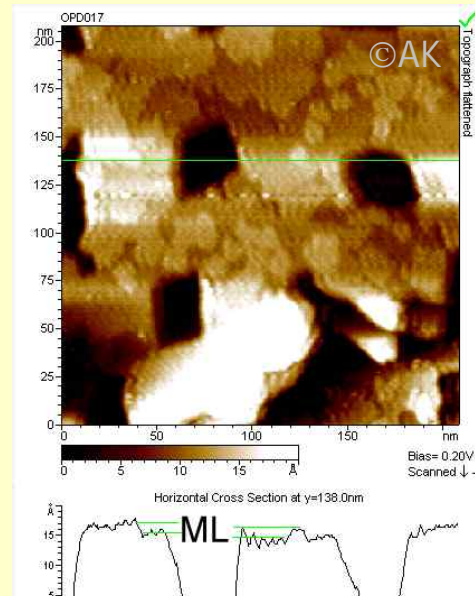
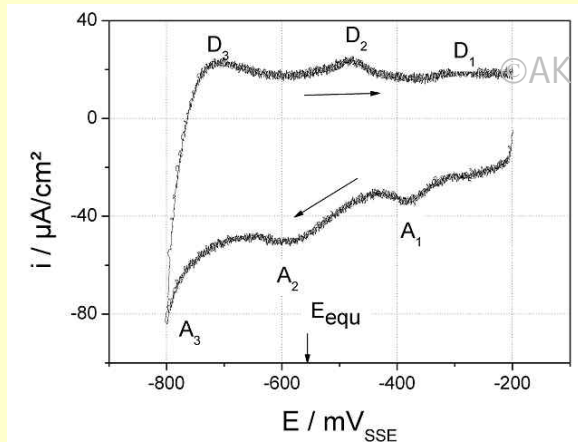


The effect of magnetic fields on the microstructure of electroplated Co layers

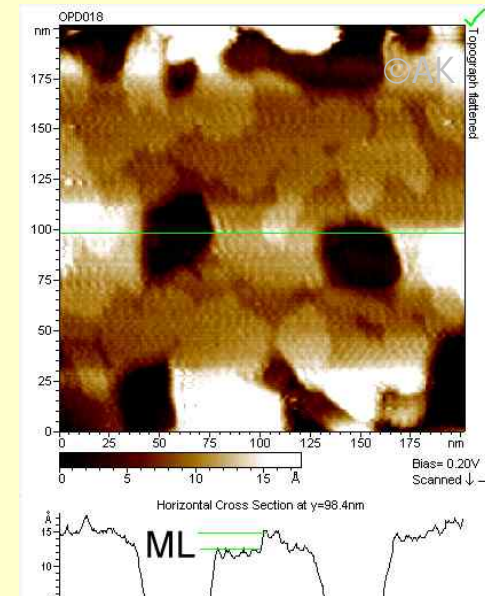
Andreas Krause and Margitta Uhlemann

- 1 Nucleation of Co on Cu(100)
- 2 Influence of Magnetic fields on mass transport
- 3 Influence of Magnetic fields on Layer Structure
- 4 Summary

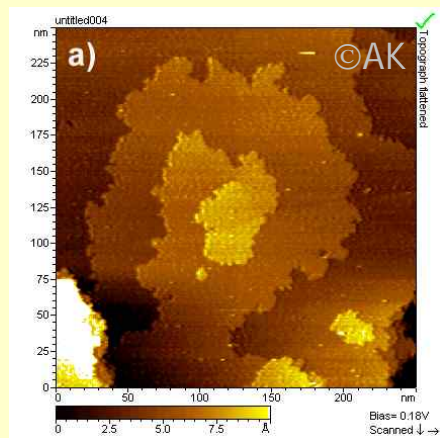
1 Nucleation of Co onto Cu(100)



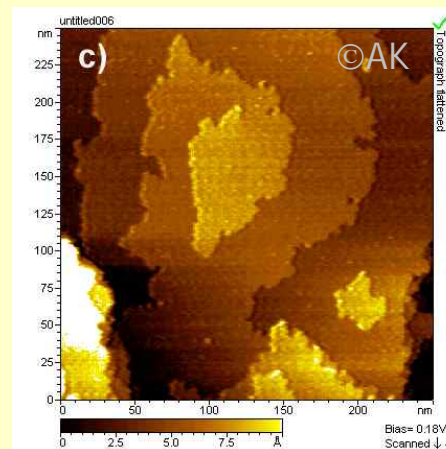
a) $E = -200 \text{ mV}_{\text{SSE}}$, $t = 0 \text{ s}$



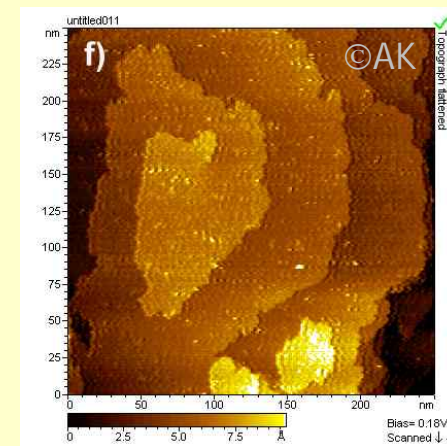
b) $E = -400 \text{ mV}_{\text{SSE}}$, $t = 90 \text{ s}$



a) $E = -200 \text{ mV}_{\text{SSE}}$, $t = 0 \text{ s}$

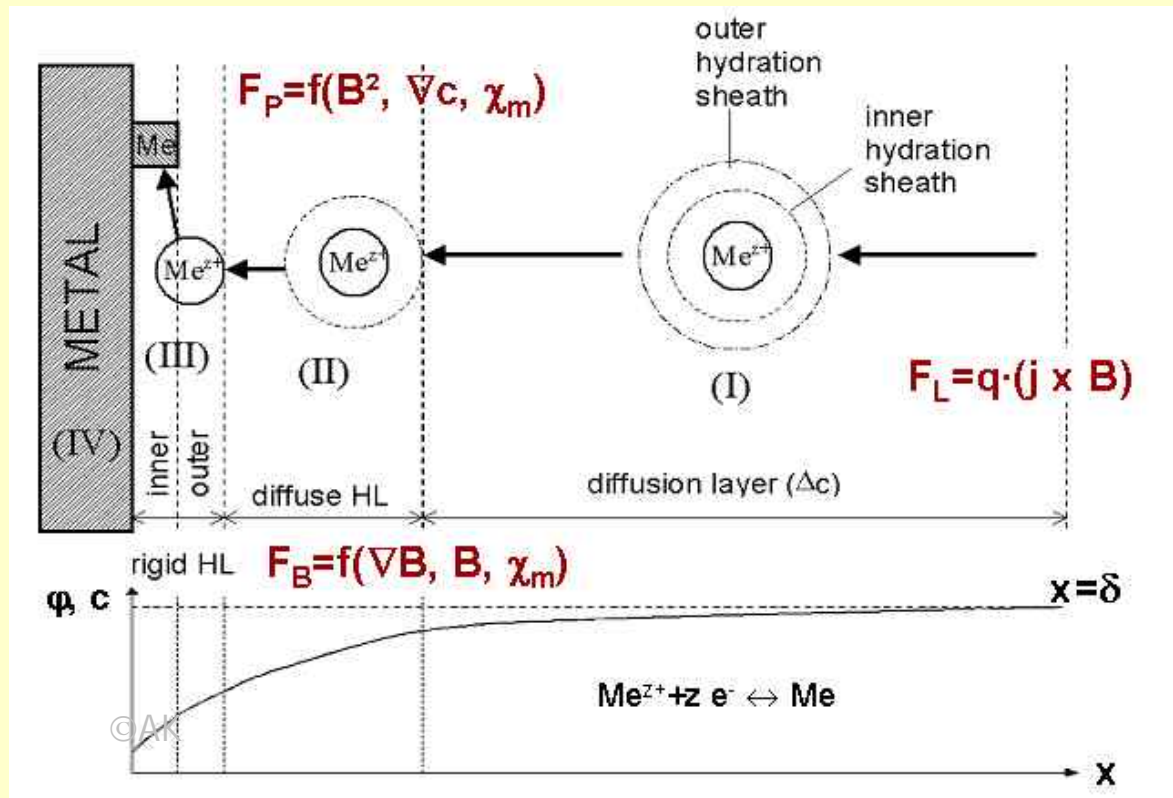


b) $E = -400 \text{ mV}_{\text{SSE}}$, $t = 170 \text{ s}$



c) $E = -600 \text{ mV}_{\text{SSE}}$, $t = 600 \text{ s}^2$

2 Influence of a magnetic field on ion transport



- **Diffusion force:**
à main driving force of the metal ions:

$$\vec{F}_D = -R \cdot T \cdot \nabla c_{(x,t)}$$

- **Lorentzforce (MHD):**
à $B \parallel$ surface: agitation reduces δ_D , $\uparrow i_D$:

$$\vec{F}_L = \vec{i} \times \vec{B}$$

$$\epsilon_{\text{mag}} = \frac{\chi_m}{2\mu_0} \cdot B^2$$

- **Field gradient force:**

à in inhomogeneous B

$$\vec{F}_{\nabla B} = \chi_m \cdot \frac{c \cdot B}{\mu_0} \cdot \nabla B$$

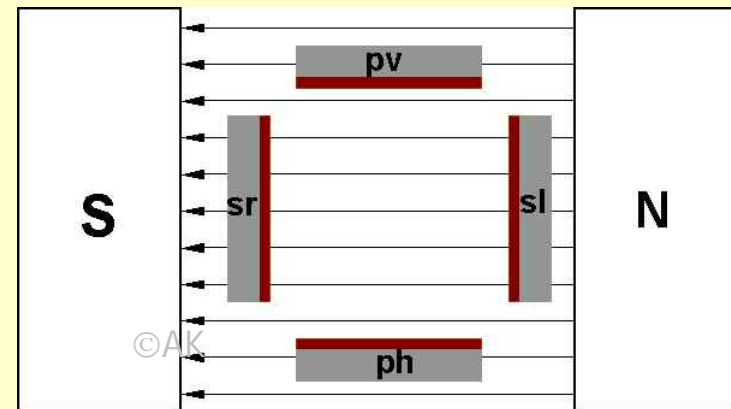
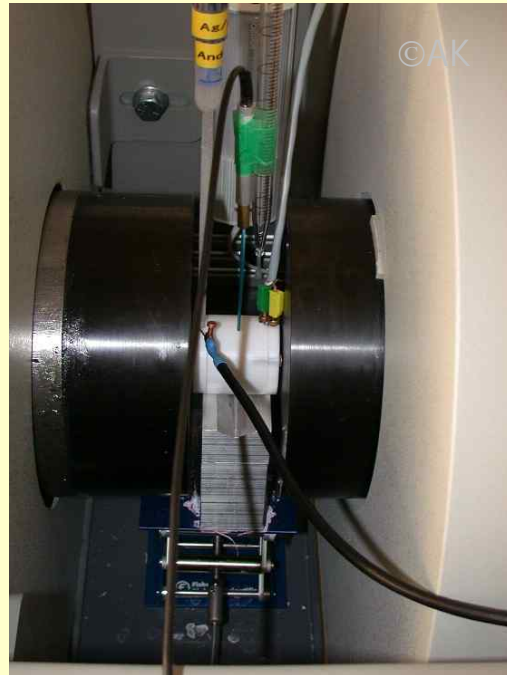
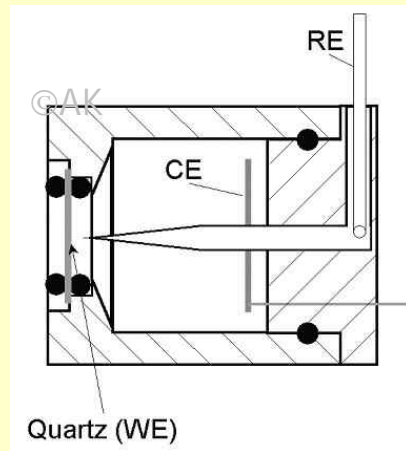
- **Paramagnetic ∇ -force:**

à in the diffusion layer (∇c)

$$\vec{F}_p = \chi_m \cdot \frac{B^2}{2\mu_0} \cdot \nabla c$$

	$\chi_m / \text{m}^3/\text{mol}$
Co^{2+}	$1 \cdot 10^8$
Cu^{2+}	$1,46 \cdot 10^7$

2 Experimental



u QMB-Cell:

- simultaneous measurement of Δm and i
- simple and constant geometry

$B = 1.2\text{T}$

u Four different magnetic field to electrode configurations

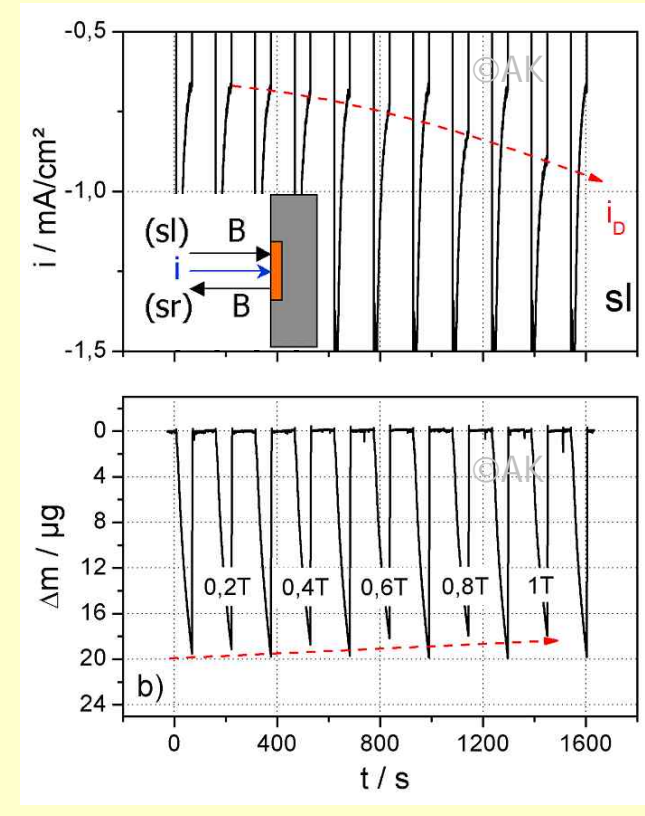
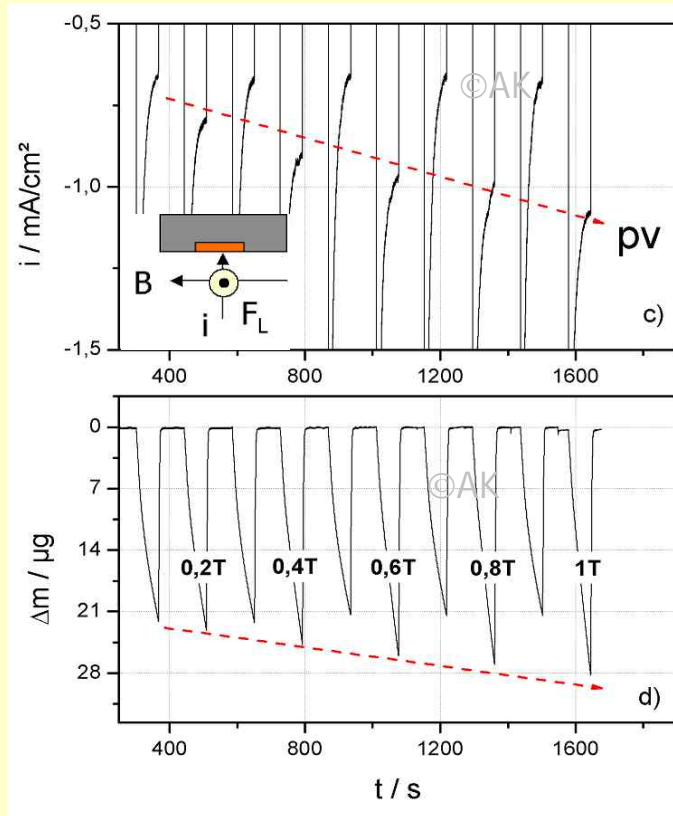
Electrolyte: 0.01M CoSO_4 , $0.1\text{M Na}_2\text{SO}_4$, pH3

2 Influence of magnetic fields on mass transport

$B \perp I$:

$E = -1V_{SSE}$, $t = 60s$

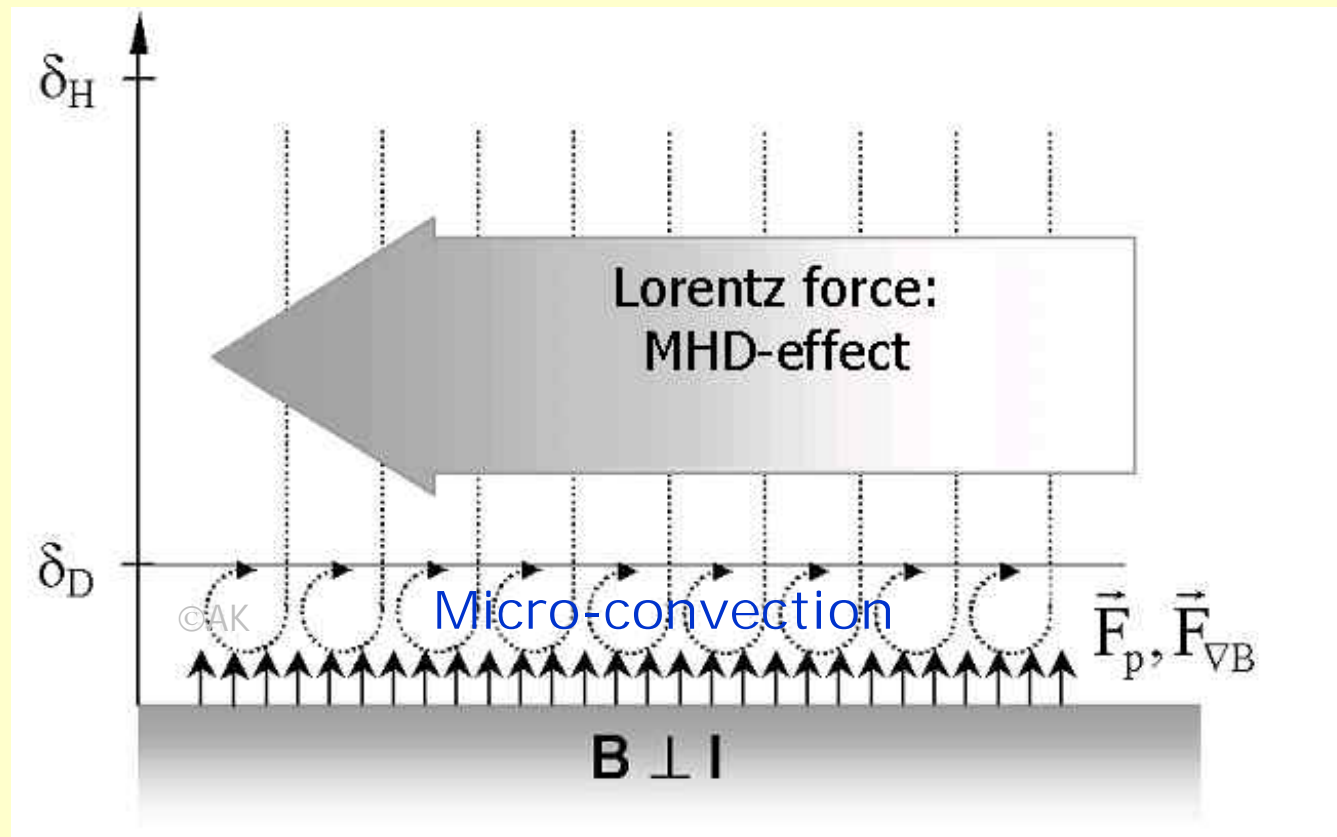
$B \parallel I$:



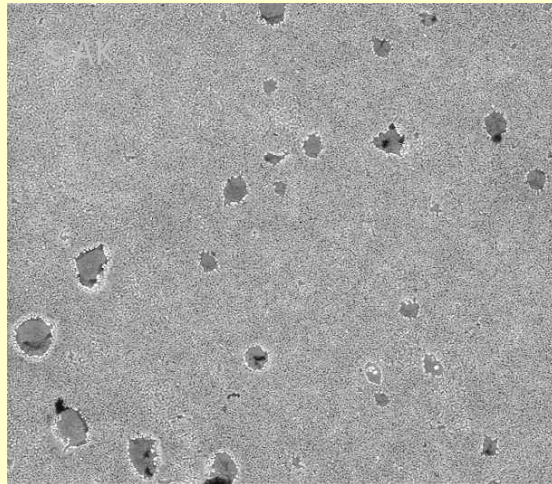
- $\uparrow |\Delta m|$, $|i_D| \sim B^{1/3} \rightarrow$ MHD-effect
- increasing current efficiency (especially for $\uparrow \eta$) \rightarrow cobalt deposition stronger affected than hydrogen reaction

- no F_L (because $B \parallel i$)
- $\downarrow |\Delta m| \rightarrow$ decreasing deposition rate due to F_p (Co-ions away from electrode)¹
- $\uparrow |i| \rightarrow B$ support the hydrogen reaction \rightarrow decreasing current efficiency! 5

Magnetic induced convection - model

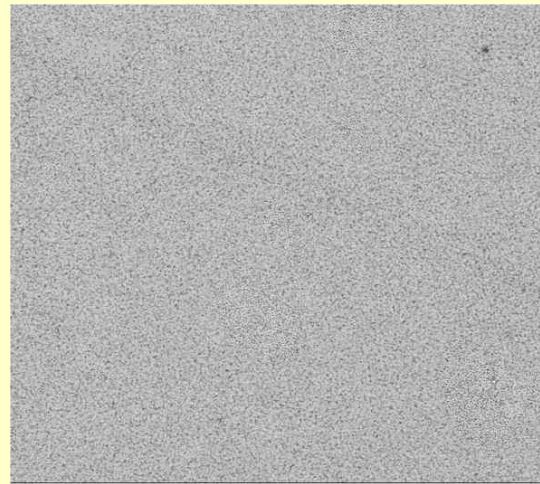


3 Influence of magnetic fields on morphology of Co-deposits



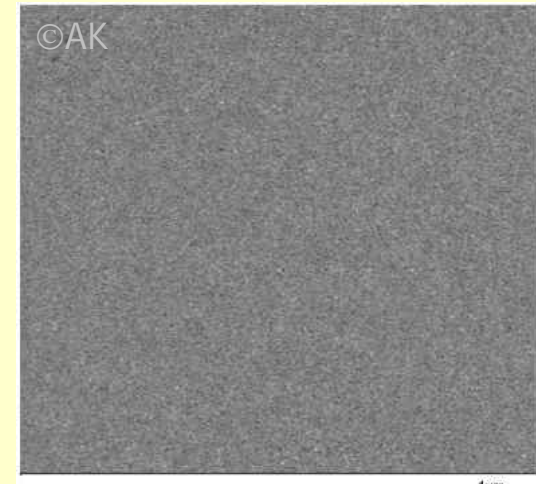
a) E3, 0T d=10nm

Mag = 20.00 K X $1\mu\text{m}$



b) E3, 1Tph d=10nm

Mag = 20.00 K X $1\mu\text{m}$



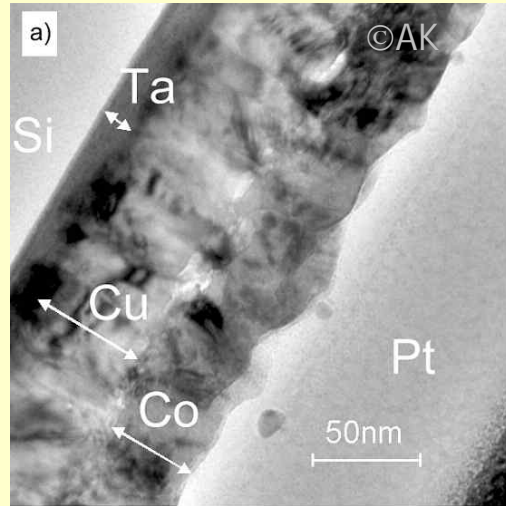
c) E3, 1T-sr d=10nm

Mag = 20.00 K X $1\mu\text{m}$

⊃ **$B=0\text{T}$, $E_3=-1100\text{mV}_{\text{SSE}}$** :
holes appear due to adsorbed hydrogen;
diameter $<1\mu\text{m}$ à inside the diffusion layer
($\delta_D \sim 100\mu\text{m}$)

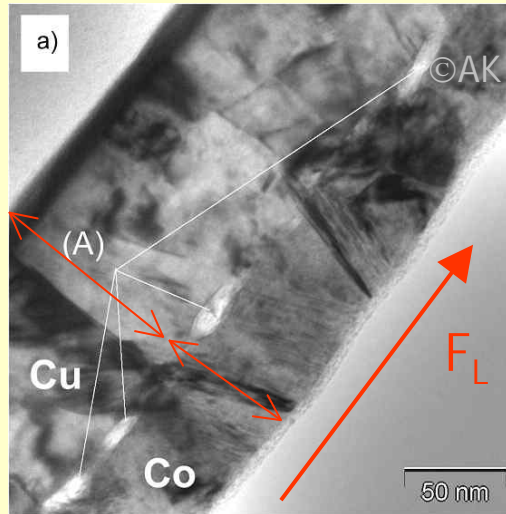
⊃ **$B=1\text{T}$ (independent on orientation),
 $E_3=-1100\text{mV}_{\text{SSE}}$** :
very smooth and closed layer, desorption
of H_2 is supported by B due to diamagnetic
effect of H_2 or magnetoconvection

Influence of magnetic fields on the microstructure of Co deposits



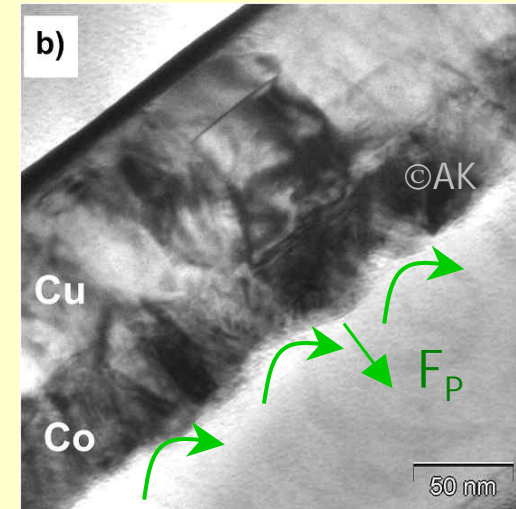
B=0:

- u irregular growth
- u high roughness
- u reaction layer at the Co-Cu interface



B || surface:

- u coherent growth
- u fibre texture of Co-layer
- u low roughness
- u reaction layer at the Co-Cu interface



B ⊥ surface:

- u disturbed microstructure
- u fibre texture of Co-layer
- u higher roughness than B || surface
- u wavy surface

3 Effects of magnetic fields on crystallographic orientation and phase formation

Free surface energy:
(dominates texture)

Co	fcc(100)	fcc(111)	hcp(100)	hcp(002)
$\gamma_{hkl} \text{ J/m}^2$	2,78	2,7	3,035	2,775

Magneto-crystalline anisotropy energy:

$$u_a = K_0 + K_1 \sin^2 \theta + K_2 \sin^4 \theta \dots$$

hcp_{max}: (100)-(002)

$$u_a = 560 \text{ kJ/m}^3$$

fcc_{max}: (111)-(100)

$$u_a = 16 \text{ kJ/m}^3$$

Surface anisotropy energy:

$$\gamma_{hkl-uvw} = (\gamma_{hkl} - \gamma_{uvw}) \cdot \sin \theta$$

crystallographic orientation:

$$d \approx \gamma_{hkl-uvw} / u_a$$

$d_{fcc} > 4 \mu\text{m}$
 $d_{hcp} > 0,46 \mu\text{m}$

Phase-formation:

Stacking fault energy:

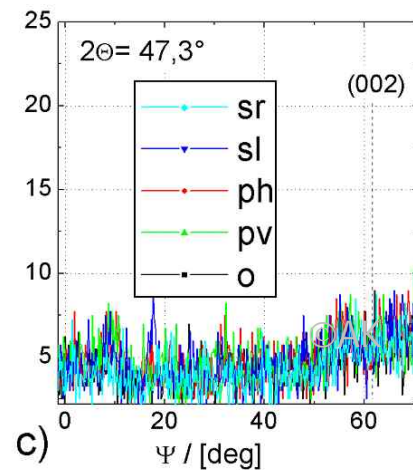
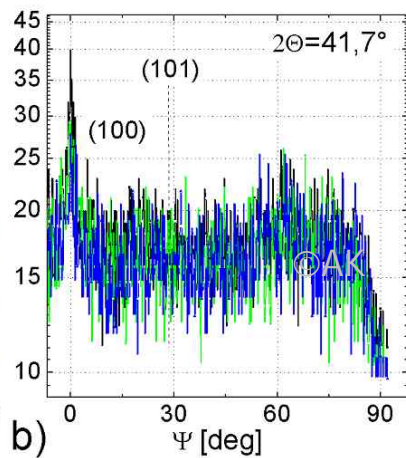
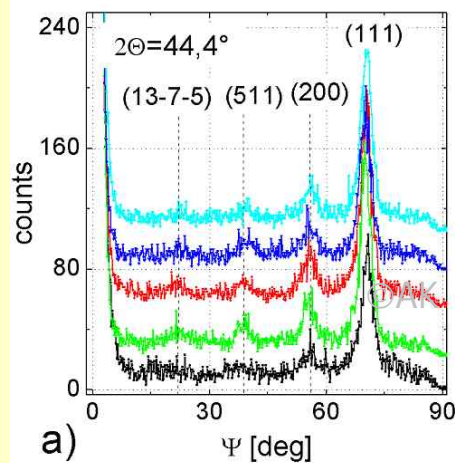
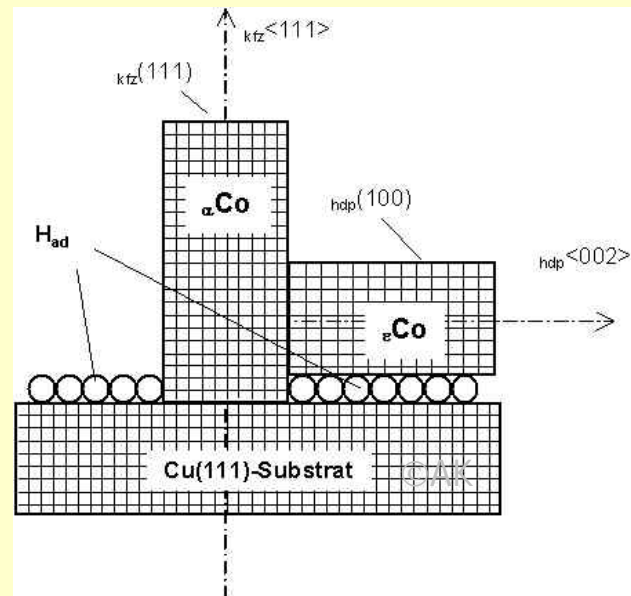
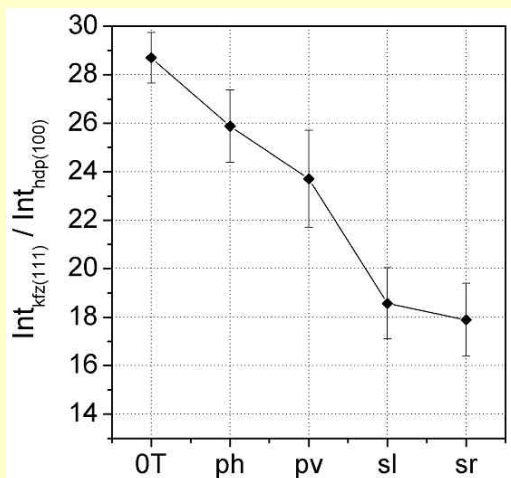
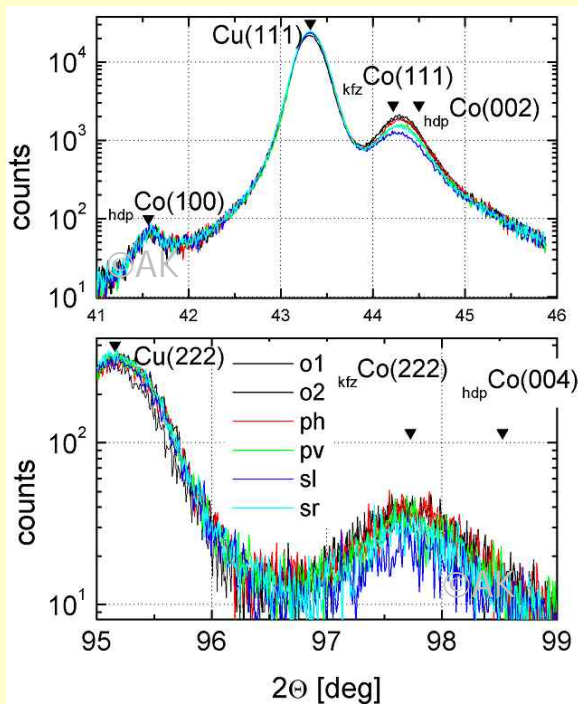
$$\gamma = 25 \text{ mJ/m}^2$$

$$B = 1 \text{ T: } \mu_0 H M \approx 10^3 \text{ kJ/m}^3$$

$$d > 25 \text{ nm}$$

3 Growth model for Co on Cu(111)

$E = -1V_{SSE}$, $d = 50\text{nm}$



Summary

Nucleation:

- 2D- growth of Co onto Cu(100)

Influence of B on ion transport:

B // electrode:

- $F_L \propto \uparrow i, \Delta m, \text{efficiency}$
- Convection in the hydrodynamic layer: MHD-effect, laminar flow at the electrode ($u=1,5 \text{ cm/s}$)

B \perp electrode:

- $F_p \propto \uparrow i$ but $\downarrow \Delta m, \downarrow \text{efficiency}$
- Magneto-convection in the diffusion layer

Influence of B on microstructure and morphology:

- hydrogen holes are reduced due to B
- more hcp-phase due to B
- less disturbed microstructure if B is parallel